# Non-equilibrium Thermodynamics of Lithium-Ion Batteries

#### Martin Z. Bazant

Department of Chemical Engineering (2008-)

Department of Mathematics (1998-)

MIT

Postdocs: Gogi Singh (MIT Math, 2005-07), Dan Cogswell (MIT ChE, 2010-)

PhD Students: Damian Burch (MIT Math '09), Liam Stanton (Northwestern, Math '09),

Todd Ferguson, Yeqing Fu (CheE), Matthew Pinson (Physics), Peng Bai (Tsinghua)

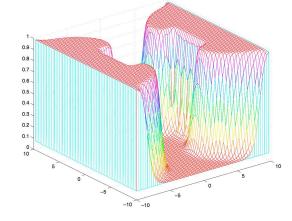
Undergraduates: Ben Derrett (U. Cambridge), Hoyin Au (Bunker Hill Community College)

Collaborators: Gerbrand Ceder (MIT, Materials)

Katsuyo Thornton (Michigan, Materials)

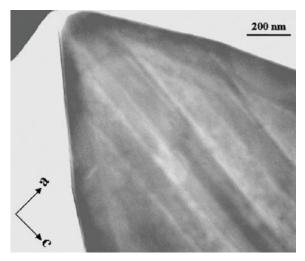
Funding: National Science Foundation DMS-084250 (Focused Research Group),

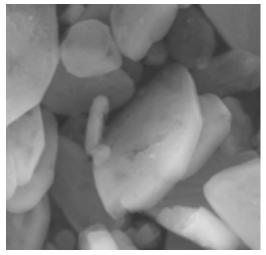
DMS-692006, DMR 02-12383, MIT Energy Initiative



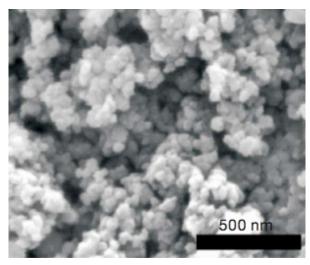
#### Outline

- 1. Motivation: Li<sub>x</sub>FePO<sub>4</sub>
- 2. Theory of ion intercalation dynamics
- 3. Phase-transformation waves
- 4. Composite electrodes





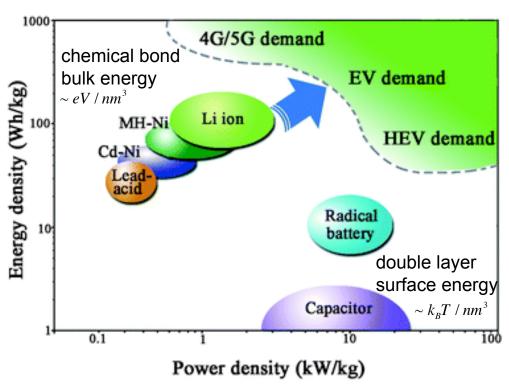




Kang & Ceder, Nature Materials (2009)

## A Grand Challenge

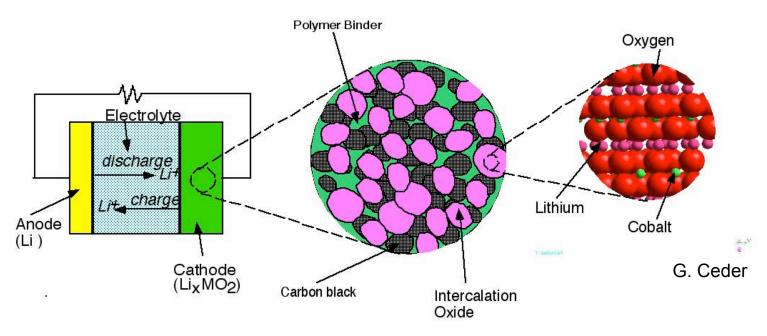
- New applications require better performance (>10x)
- Energy density is nearing theoretical limits (<10x)</li>
- Power density can improve (>>10x?), but must not sacrifice capacity or cycle life
- Predictive mathematical models are needed to interpret data & guide engineering



"Ragone plot" for electrochemical energy storage

Gao & Yang, Energy Environ. Sci. (2010)

### Multiscale Li-ion Battery Physics



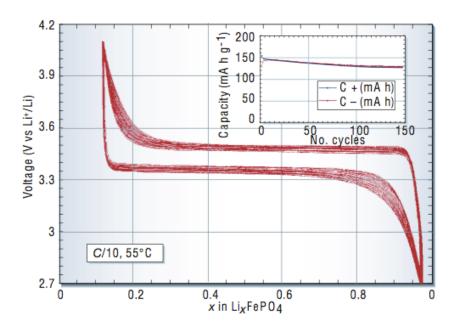
- 1. macroscopic 2. microscopic 3. atomic

- 1. Porous electrode theory (J. Newman, Berkeley)
- 2. ??? (this work)
- 3. Quantum simulations (G. Ceder, MIT)

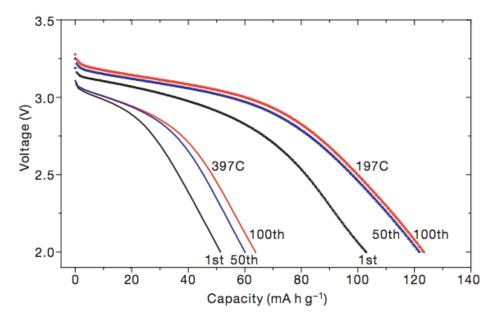
# Li<sub>x</sub>FePO<sub>4</sub>

- Advantages: stable, non-toxic, inexpensive, high-rate capability
- "Ultrafast" discharge possible (10 sec!)
- Phase separating (voltage plateau)





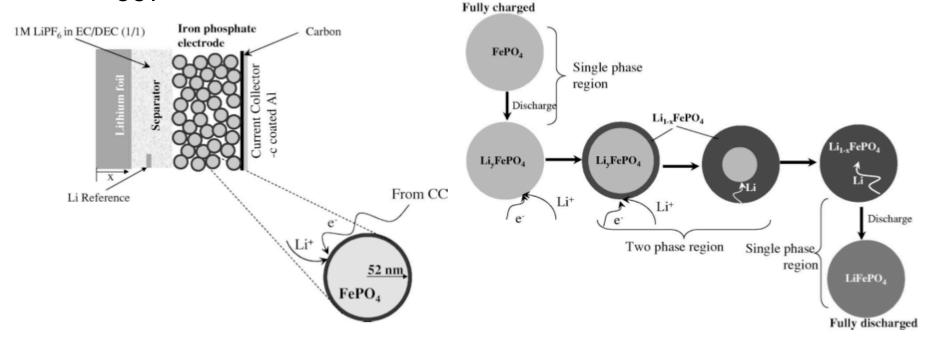




Kang & Ceder, Nature, 2009

# Porous Electrode Theory

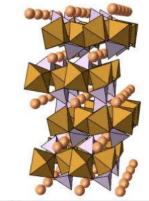
- Formal volume averaging, overlapping continua for electrolyte (ions) and electrode (electrons)
  - Tobais & Newman (1963), Newman et al 1970s
- Li-ion intercalation: Isotropic diffusion in solid spheres
  - Doyle, Fuller, Newman (1993),
- Li<sub>x</sub>FePO<sub>4</sub>: "shrinking core model" Srinivasan & Newman (2004)
- V<sub>OCV</sub>(x), D(c), stable compositions = adjustable params

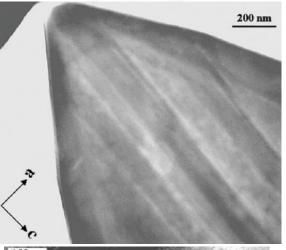


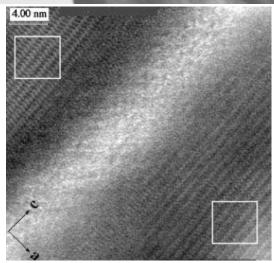
# Surprises in Li<sub>x</sub>FePO<sub>4</sub>

- Atomistic simulations: (Morgan, van der Ven, Ceder, 2004)
  - Fast Li transport in 1D channels
  - Slow 2D electron transport
- Experiments: (Chen et al 2006, Laffont et al 2006, chemical lithiation)
  - No shrinking core
  - Nano-scale phase boundaries,
     aligned with FePO<sub>4</sub> planes
  - Apparent motion along active facet, perpendicular to Li flux

Mathematical model???

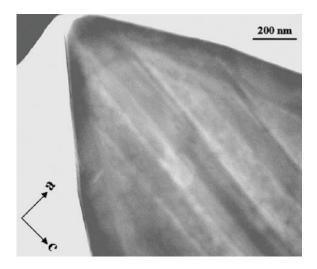


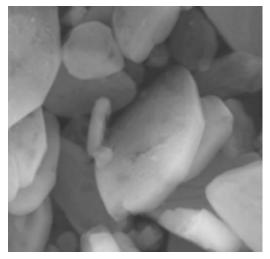




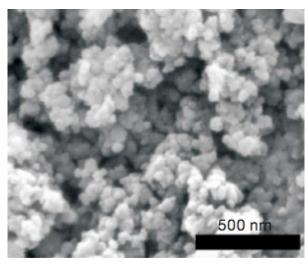
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Chen, Song & Richardson, Electrochem. Sol. State Lett. (2006)



Kang & Ceder, Nature Materials (2009)

# Modeling Strategy

Singh, Ceder, MZB, *Electrochimica Acta* (2008) Burch & MZB, *Nano Letters* (2009) MZB, *Electrochemical Energy Systems*, MIT course notes (2009-10)

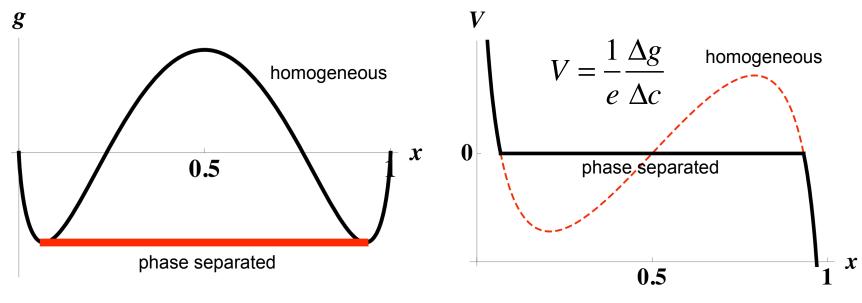
- Focus on basic physics: scalings, few params
- Self-consistently predict V<sub>OCV</sub>(Q,T), D(c,T), stable compositions, phase transformations...
- Derive nonlinear equations ("beyond 10.50")
- Incorporate randomness nucleation, porous microstructure,...
- Integrate transport and Faradaic reactions with non-equilibrium thermodynamics

## Equilibrium Thermodynamics

Homogeneous free energy density, e.g. regular solution model:

$$\overline{g}(c) = V^{\Theta}c + ac(1-c) + kT[c\log c + (1-c)\log(1-c)]$$

enthalpy density standard entropy density = potential + (particle-hole repulsion) + (ideal mixture of particles and holes)



$$if \quad T < T_c = \frac{\alpha}{k}$$

common tangent if  $T < T_c = \frac{a}{k_R}$  voltage plateau

## Non-equilibrium Thermodynamics

Singh, Ceder, MZB, Electrochimica Acta (2008)

Total free energy (solid phase)

$$G = \int \left( \overline{g}(c) + \frac{1}{2} \nabla c \cdot K \nabla c + \frac{1}{2} \sigma : \varepsilon + \dots \right) dV + \oint n \cdot \gamma(c) da$$

Bulk (diffusional) chemical potential

$$\mu \equiv \frac{\delta G}{\delta c} = \overline{g}'(c) - \nabla \cdot K \nabla c + U : \sigma + \dots$$
homogeneous gradient lattice mismatch echem. potential + penalty + strain energy

Cahn-Hilliard Equation (1958)

$$\frac{\partial c}{\partial t} + \nabla \cdot F = 0, \quad F = -Mc\nabla \mu$$

Variational boundary condition

$$n \cdot (K\nabla c + \gamma'(c)) = 0$$

Intercalation boundary condition (new)

$$n \cdot F = R(c, \mu, \mu_e)$$

### Reactions in Concentrated Solutions

MZB, Electrochemical Energy Systems, MIT course notes (2009-10) + papers to follow...

Chemical potentials  $\mu = k_B T \ln a = k_B T \ln(\gamma c) = k_B T \ln c + \mu^{ex}$ 

Reaction rate: state 1 ←(activated state)→ state 2

$$R = R_0 \left( c_1 e^{(\mu_a^{ex} - \mu_1^{ex})/kT} - c_2 e^{(\mu_a^{ex} - \mu_2^{ex})/kT} \right) = \frac{R_0}{\gamma_a} (a_1 - a_2)$$

Basic idea: the reaction complex diffuses over an activation barrier (>>kT) between two local minima (states 1,2) in a *landscape of excess chemical potential* 

Note: we cannot simply "replace concentrations with activities"

### **Equilibrium Voltage**

Faradaic reaction: reduced state  $\leftarrow \rightarrow$  oxidized state + n e<sup>-1</sup>

Chemical potentials  $\mu_1 = \mu_R = k_B T \ln(\gamma_R c_R) + q_R \phi$ 

$$\mu_2 = \mu_O + n\mu_{e^-} = k_B T \ln(\gamma_R c_R) + q_O \phi - ne\phi_e$$

Charge conservation  $q_R = q_O - ne$ 

Interfacial voltage  $\Delta \phi = \phi_e - \phi$   $\Delta \phi = \Delta \phi_{eq} \Leftrightarrow \mu_1 = \mu_2$ 

Nernst equation  $\Delta \phi_{eq} = \frac{k_B T}{ne} \ln \frac{a_O}{a_R}$ 

# Theory of Electrochemical Reactions in Concentrated Solutions

#### "Butler-Volmer hypothesis":

electrostatic energy of transition state = weighted average of states 1 and 2

$$\mu_a = k_B T \ln \gamma_a + \alpha q_R \phi + (1 - \alpha)(q_O \phi - ne \phi_e)$$

Overpotential

$$\eta = \Delta \phi - \Delta \phi_{eq} = \mu_2 - \mu_1 = \Delta \mu$$

**Butler-Volmer equation** 

$$J = neR = J_0 \left( e^{(1-\alpha)ne\eta/kT} - e^{-\alpha ne\eta/kT} \right)$$

Exchange current density

$$J_{0} = neR_{0}c_{R}^{\alpha}c_{O}^{1-\alpha}\left(\frac{\gamma_{R}^{\alpha}\gamma_{O}^{1-\alpha}}{\gamma_{a}}\right)$$

Note: we recover classical Butler-Volmer kinetics only in a dilute solution, where all activity coefficients = 1

### Models for Ion Intercalation Kinetics

**Example:** Lithium intercalation

$$Li_{(s)} \leftarrow \rightarrow Li^+_{(l)} + e^-$$

$$\begin{split} \mu_1 &= \mu_{Li(s)} = k_B T \ln \left( \frac{c}{1-c} \right) + a(1-2c) - \nabla \cdot K \nabla c \\ \mu_2 &= \mu_{Li^+(l)} + \mu_{e^-} = k_B T \ln c_+ + e(\phi - \phi_e) = \mu_{ext} \\ \gamma_a &= \frac{1}{1-c} \quad \text{only excluded volume} \\ &\text{in transition state} \end{split}$$

Cahn-Hilliard in the solid phase

Symmetric Butler-Volmer

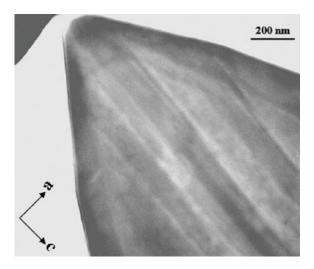
$$J = 2J_o \sinh\left(\frac{e(\mu_2 - \mu_1)}{2k_B T}\right)$$

$$J_0 = eR_0 \sqrt{c_+ c(1-c)} \exp\left(\frac{-2ac - \nabla \cdot K \nabla c}{2k_B T}\right)$$

Reaction rate depends strongly on concentration... and on concentration gradients!

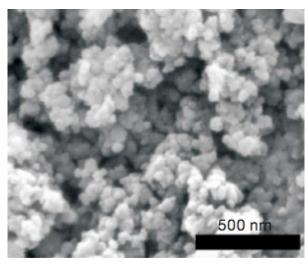
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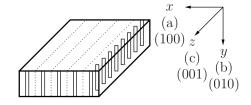


Chen, Song & Richardson, Electrochem. Sol. State Lett. (2006)



Kang & Ceder, Nature Materials (2009)

# Reaction-limited Intercalation in Anisotropic Nanocrystals



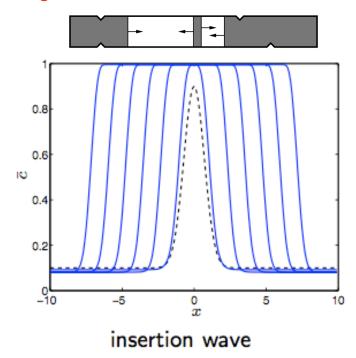
- Assume for Li<sub>x</sub>FePO<sub>4</sub>
  - no phase separation, fast diffusion in y direction (Da<sub>x</sub><<1)</li>
  - strong phase separation, slow diffusion ( $Da_{x,z} << 1$ )
- 2D nonlinear "forced Allen-Cahn" equation

$$\frac{\partial c}{\partial t} = R(c, \nabla \cdot K \nabla c) \propto \sinh(\mu_{ext} - \mu)$$

- Predicts travelling waves, instabilities...
- (This new model may also describe electrophoretic deposition, nanoparticle adsorption in double layers...)

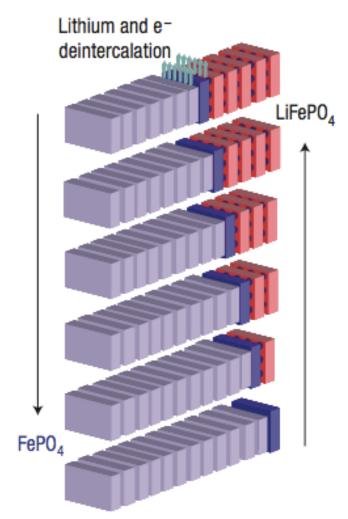
### Prediction: Intercalation Waves

Singh, Ceder, MZB, Electrochimica Acta (2008)



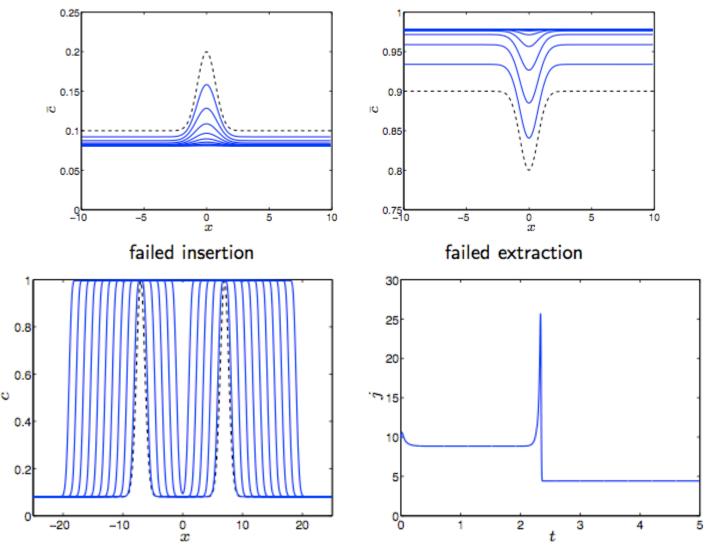
$$\frac{\partial c}{\partial t} = R(c, \nabla^2 c; \mu_{ext})$$
 Nonlinear wave equation

- layer-by-layer insertion/extraction
- reaction limited (D irrelevant!)
- nucleation required



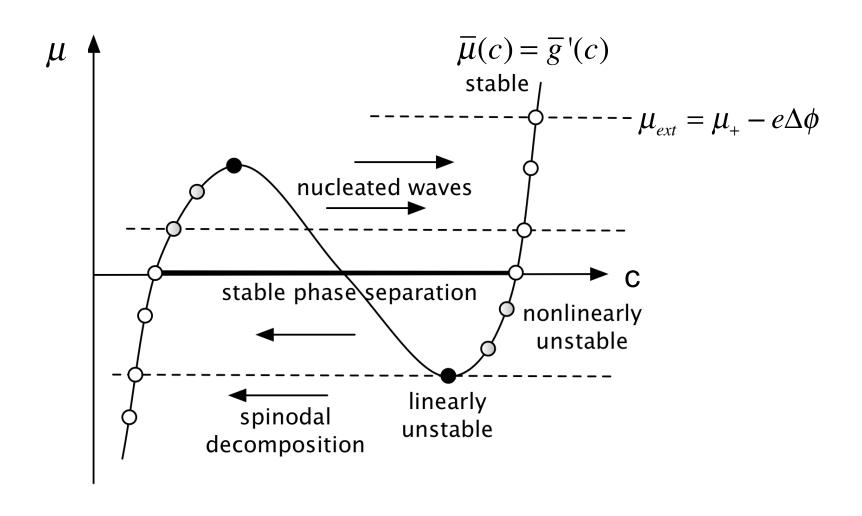
Delmas et al, *Nature Materials* (2008) "domino cascade model" (no math)

### Wave nucleation and annihilation

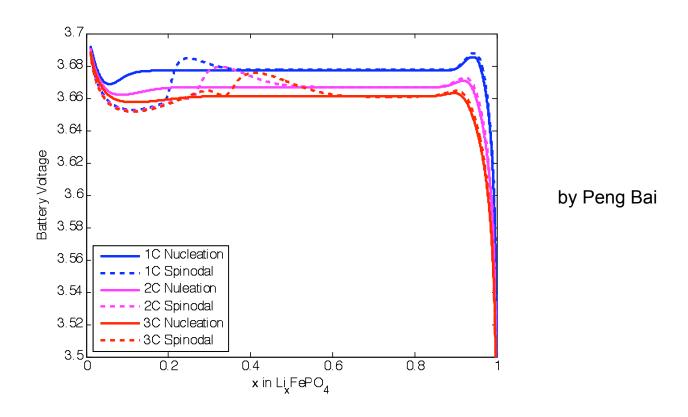


Waves collide at constant voltage ...and produce a current spike + step

# Theory of intercalation phase transformations at constant voltage



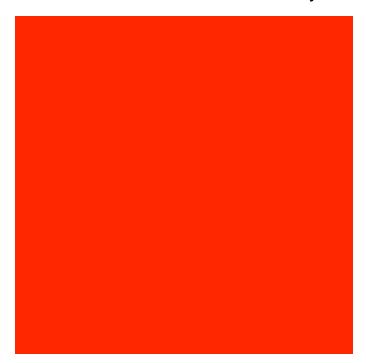
# Prediction: Overshoot of the voltage plateau at constant current



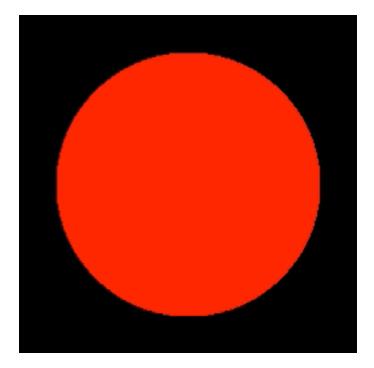
- Defects or facet edges can act as nucleation sites for waves
- In a perfect crystal, phase transformation is triggered by linear instability (like spinodal decomposition, but in an open system)

# Simulations of 2D reaction-limited phase transformations at constant current

by Dan Cogswell



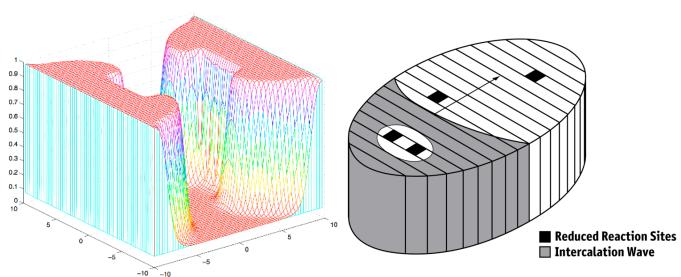
Linear instability in a perfect crystal

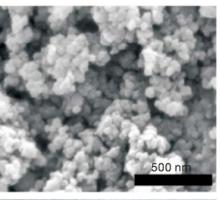


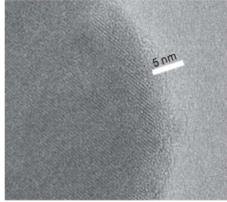
Stochastic nucleation at the facet edge

### Wave-defect interactions

Burch, Singh, Ceder, Bazant, Solid State Phenomena (2008)



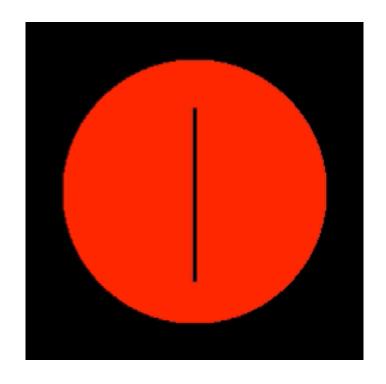


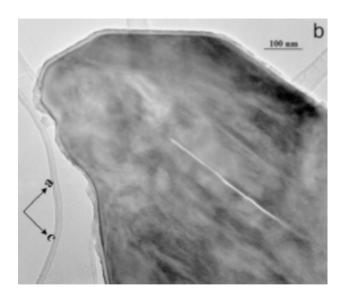


- "kinetic defect" = reduced reaction rate
- inter-phasial tension & elasticity hinder wave motion
- new mechanism for capacity fade & power loss
- may explain ultrafast discharging due to 5nm phosphate glass flim on LiFePO4 nanoparticles (Kang & Ceder 2009)

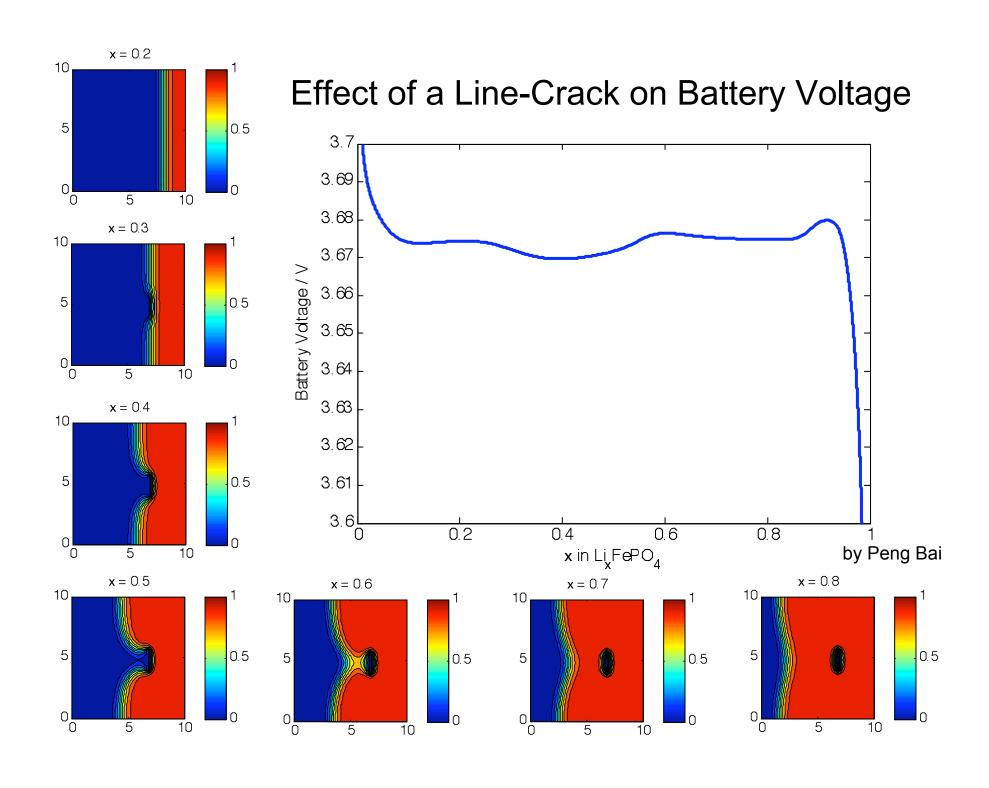
# Intercalation at constant current in the presence of a crack

by Dan Cogswell





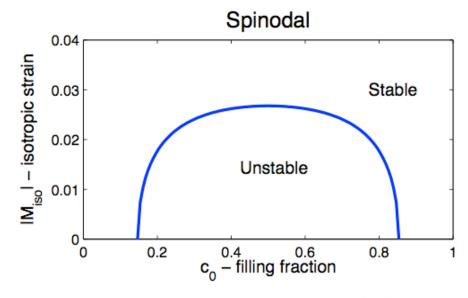
Chen, Song & Richardson, Electrochem. Sol. State Lett. (2006)



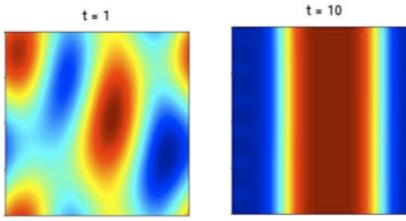
### Elastic strain effects

by Liam Stanton

 Elastic energy from lattice mismatch strain suppresses phase separation

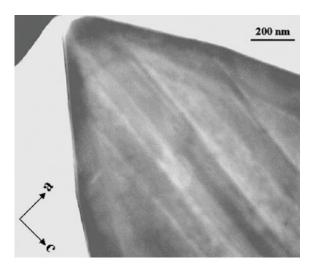


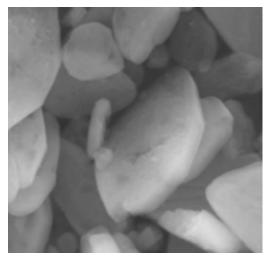
 Phase boundary aligns with most rigid planes (softest crystal axis, maximum strain)



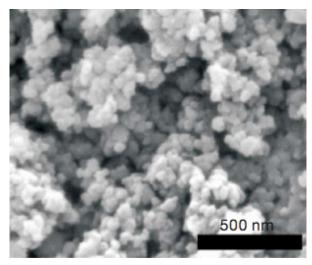
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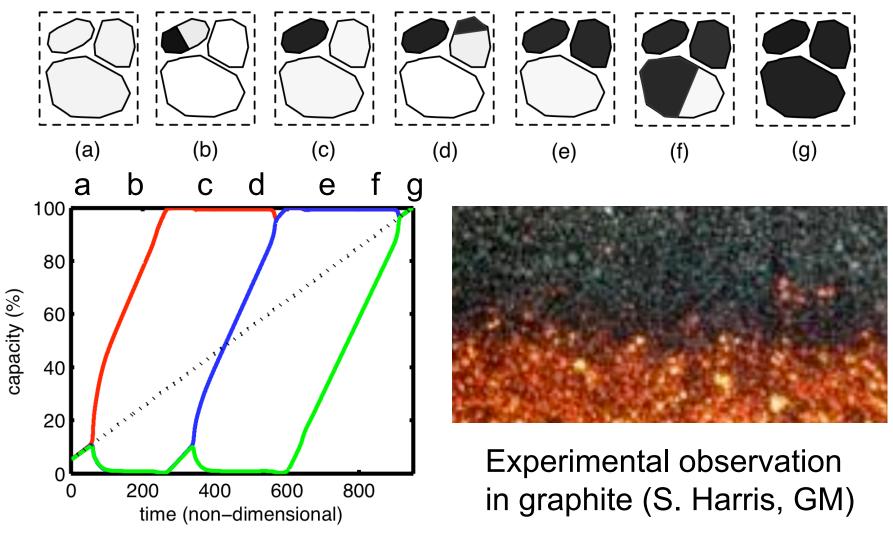
Chen, Song & Richardson, Electrochem. Sol. State Lett. (2006)



Kang & Ceder, Nature Materials (2009)

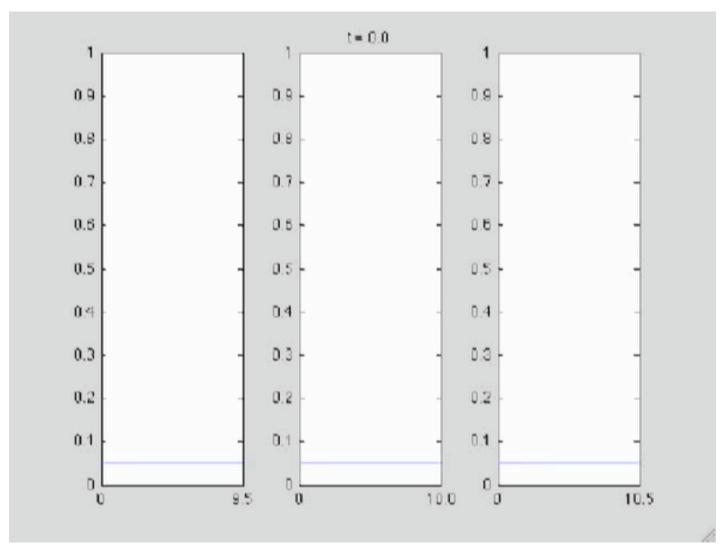
# "Mosaic Instability"

Burch, Ceder, Bazant, in preparation



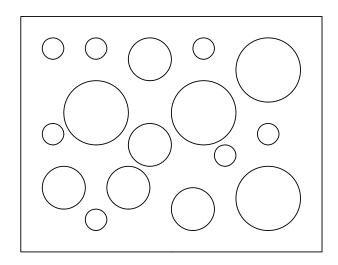
Chem. Phys. Lett. (2010)

# Mosaic Instability



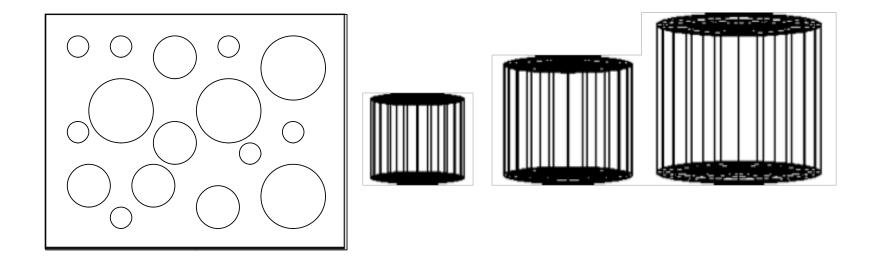
Concentration profiles of intercalated Li in three particles at constant total current (D. Burch)

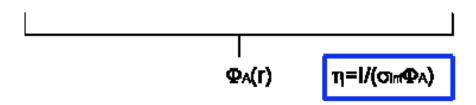
analysis by Matthew Pinson

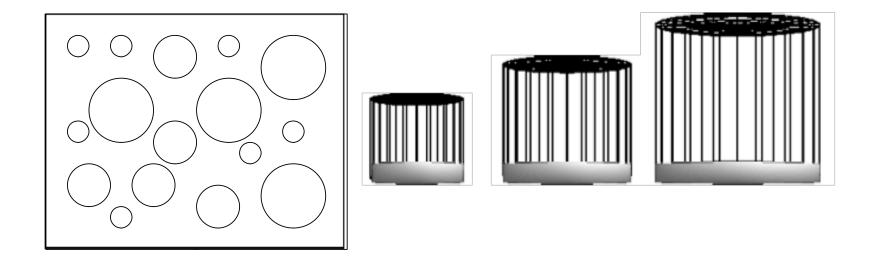


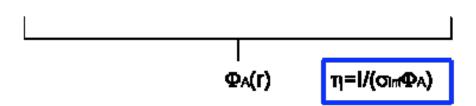
Assume each particle intercalates at constant voltage (due to phase separation) until it is full.

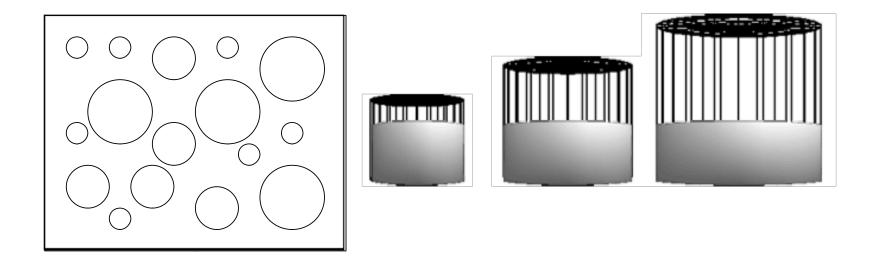
At constant total current, as the number of active particles decreases, the cell voltage decreases – analogous to the fiber bundle model for mechanical failure.

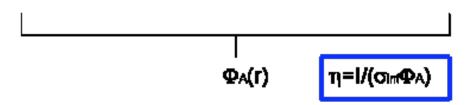


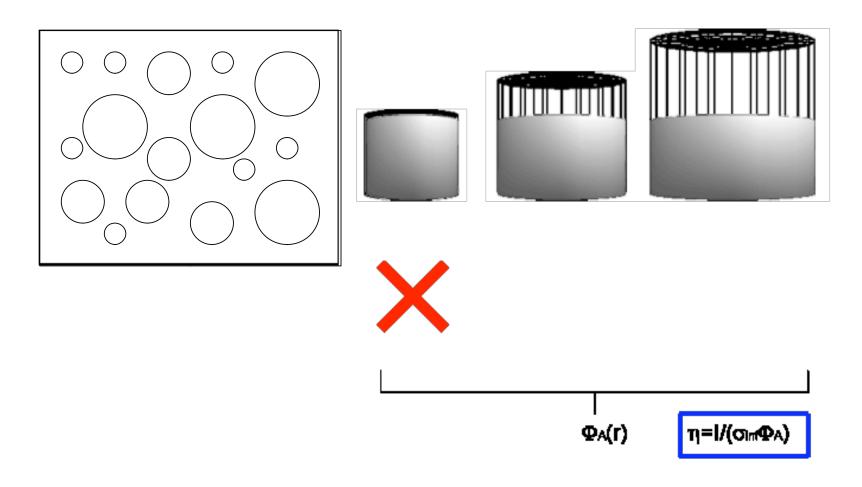


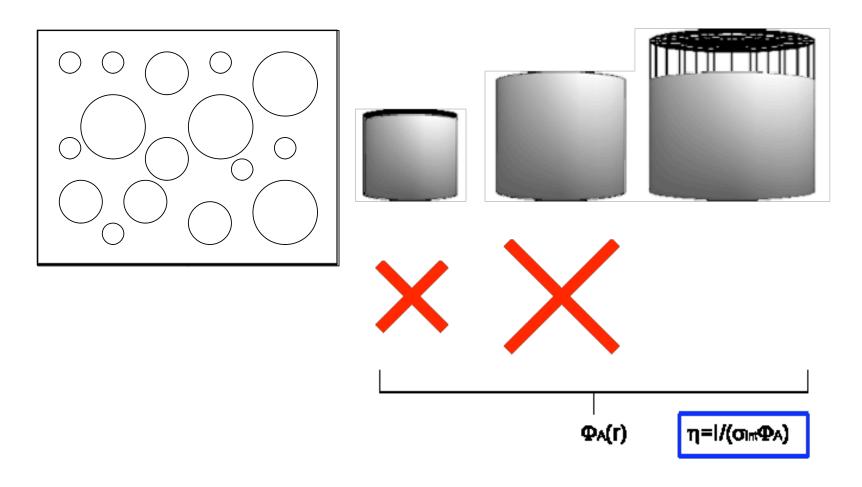


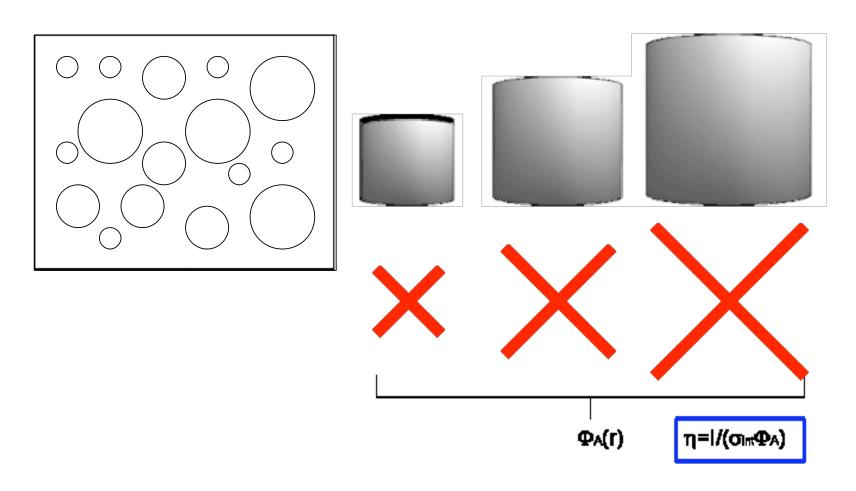




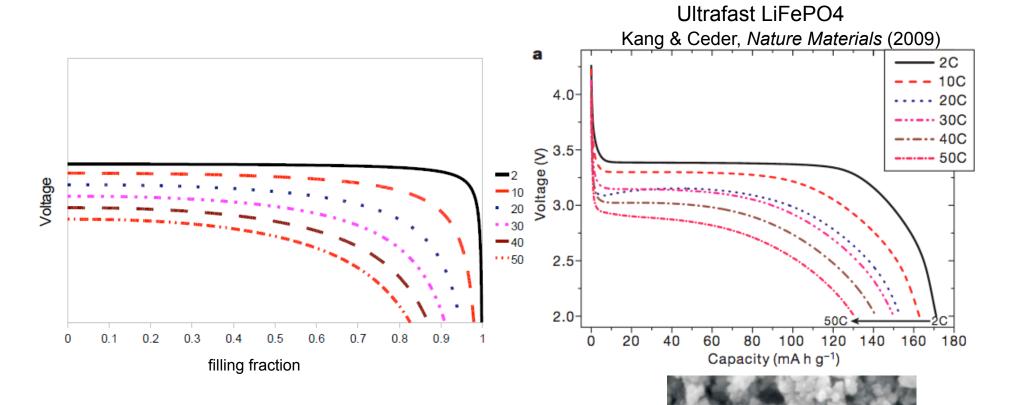








# Comparison with experiment



Theory with only one adjustable parameter (interfacial resistance), and a broad size distribution

# Summary

- Nanoscale modeling framework for Li-ion batteries based on non-equilibrium thermodynamics
- Reaction-limited insertion waves
- Mosaic instability
- "Fiber-bundle effect"
- Other work in the group:
  - Modified porous electrode theory, fit to experiments
  - Double layers (Frumkin correction, capacitive charging)
  - Stochastic effects (capacity fade, lifetime)

http://bazantgroup.mit.edu/BMG