#### FINAL EXAM REVIEW:

Material covered—Mohr's Circle until end. The exam covers recitations 8-12. (only 5 recitations!)

#### TRANSFORMAION OF STRESS & STRAIN

# Stress transformation formulas for plane stress at a given location MOHR's CIRCLE

MOTH S CIRCLE
$$\begin{aligned}
\sigma_{\chi'} - \frac{\sigma_{\chi} + \sigma_{y}}{2} &= \frac{\sigma_{\chi} - \sigma_{y}}{2} \cos 2\theta + \tau_{\chi y} \sin 2\theta \\
\tau_{\chi' y'} &= -\frac{\sigma_{\chi} - \sigma_{y}}{2} \sin 2\theta + \tau_{\chi y} \cos 2\theta \\
\left(\sigma_{\chi'} - \frac{\sigma_{\chi} + \sigma_{y}}{2}\right)^{2} + \tau_{\chi' y'}^{2} &= \left(\frac{\sigma_{\chi} - \sigma_{y}}{2}\right)^{2} + \tau_{\chi y}^{2} \\
\sigma_{Avg} &= \frac{\sigma_{\chi} + \sigma_{y}}{2} \\
R &= \sqrt{\left(\frac{\sigma_{\chi} - \sigma_{y}}{2}\right)^{2} + \tau_{\chi y}^{2}}, \text{ R is equal to the maximum shear stress} \\
\left(\sigma_{\chi'} - \sigma_{Avg}\right)^{2} + \tau_{\chi' y'}^{2} &= R^{2}
\end{aligned}$$

### **ELASTIC MODULI**

Linear elasticity--- modeling bonds between atoms- Lennard Jones Potential Review all equations from this,

(aka. Write them on your sheets)

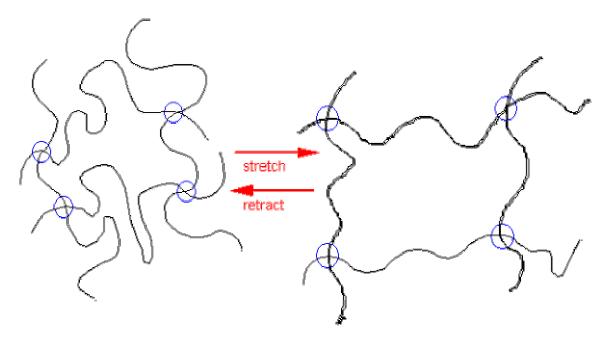
remember: re is the distance where F(r)=dU(r)/dr=0

## Rubber Elasticity—Freely Jointed Chain Model

## →a random walking polymer at finite T is a Hookean spring

Why is this useful?

Because these equations define the stretching of a single polymer chain.



Loss of entropy upon stretching, means that there is a retractive force for recovery when external stress removed.

## This is why a rubber band returns to its original shape.

A simple reminder of polymer statistics. Suppose the walk has N links: End to end distance R(N)

## From Rubber elasticity:

r = instantaneous chain end-to-end separation distance (Draw on board--- squiggly lines with beginning and end separated by r)  $\langle r^2 \rangle = na^2$  root mean square end to end distance a = statistical segment length—local chain stiffness n = # of a's Lc = contour length—length of fully extended chain.

Probability of finding a free chain end a radial distance, r, away from a fixed chain end (origin)  $\sim$  omega = P(R) =  $(4b^3r^2)/\text{sqrt}(pi)*\exp(-b^2r^2)$  where b = sqrt  $(3/(2na^2))$ 

This is Gaussian form

Macrostate is defined by the length r.

Microstates are the different random walks. So..

$$P(R) \sim e^{-3R^2/2Nb^2} \sim \Omega(R)$$
 (# of µstates with length R) (N=n and b=a)

Configurational Entropy (measure of disorder) = S = kb\*ln[P(r)]

Helmholtz Free Energy = A or H = -Tkb\*ln[P(r)]

Entropic elastic force, linear elasticity (hookean spring) f or  $F = -dA(r)/dr^2$ Entropic chain stiffness = k = dF(r)/dr or second derivative of A.

#### VISCOELASTICITY

Review Spring-Dashpot models.

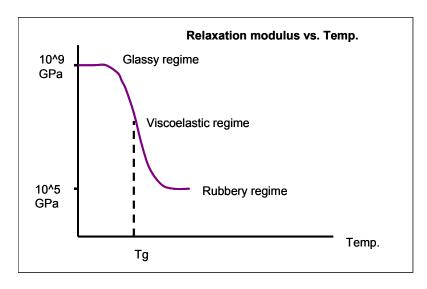
Behavior between elastic solid and viscous fluid (hence the name viscoelastic)

Creep test- Const. stress

If viscoelastic, strain proportional to stress change

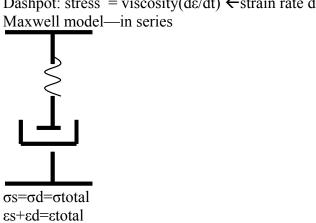
Relaxation Test - Const. Strain

Now, temperature effect on amorphous polymers



Spring: E=stress/strain

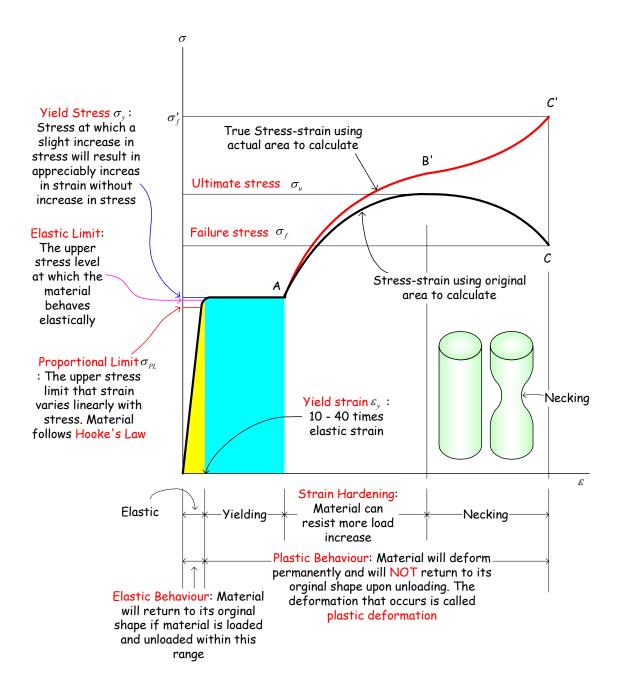
Dashpot: stress = viscosity(dɛ/dt) ← strain rate drain



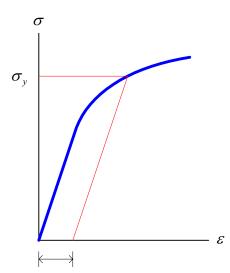
Voigt or Kelvin Model (in parallel) σs+σd=σtotal εs=εd=εtotal

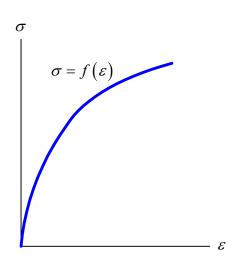
Neither give entire model of a polymer—need to sum both models together.

Next, Stress-strain diagrams—now going into plasticity!



- If structure made of ductile materials is overloaded, it will present large deformation before failing
- Some ductile materials do not exhibit a well-defined yield point, we will use offset method to define a yield strength
- Some ductile materials do not have linear relationship between stress and strain, we call them nonlinear materials





0.002 or 0.2% offset

The work done, which equals to the strain energy stored in the element, is

$$dU = \frac{1}{2}(\sigma dxdy)(\varepsilon dz)$$
 or  $dU = \frac{1}{2}\sigma\varepsilon dV$ 

The total strain energy stored in a material will be

$$U = \int_{V} \sigma \varepsilon dV$$

> The strain energy per unit volume or the strain energy density is

$$u = \frac{dU}{dV} = \frac{1}{2}\sigma\varepsilon$$

If the material is linear elastic ( $\sigma = E\varepsilon$ , Hooke's Law holds), the strain energy density will be

$$u = \frac{1}{2}\sigma\left(\frac{\sigma}{E}\right) = \frac{1}{2}\frac{\sigma^2}{E}$$

Plasticity—3-D

## **Yield Criteria: Tresca Criterion**

Sometimes called the Maximum shear stress criterion

- 1. get yield when max. shear stress, tau, in component (under general stress state) equals the ma, tau, in a uniaxial tensile test at yield
- 2. principle stresses  $\sigma_1 > \sigma_2 > \sigma_3$

You could take stress state, rotate and rewrite it. Mohr's circle does this.

For general stress state:

$$\tau_{\text{max}} = (\sigma_1 - \sigma_3)/2$$

In uniaxial test:  $\sigma_1$  ( $\sigma_2 = \sigma_3 = 0$ )

$$\tau_{\text{max}} = (\sigma_1)/2$$
  
so, you'd get yield when  $\sigma_1 = \sigma_Y$ 

In general, you get yield when  $\sigma_1 - \sigma_3 = \sigma_Y$ 

Tresca says, you take the difference between the biggest and smallest stress to get yield stress.

## **Yield Criteria: Von Mises Criterion**

You get yield when equivalent sigma = yield stress

$$\sigma_{eq} = \operatorname{sqrt}(0.5 * [(\sigma_{11} - \sigma_{22})^2 + (\sigma_{22} - \sigma_{33})^2 + (\sigma_{33} - \sigma_{11})^2] + 3\sigma_{23}^2 + 3\sigma_{13}^2 + 3\sigma_{12}^2)$$

These are looking at effects of Shear on the Material.

Good things to remember--

At R.T. many materials (esp. metals) have a well defined yield stress,  $\sigma_v$ 

 $\sigma < \sigma_v$  elastic recoverable deformation

 $\sigma > \sigma_v$  plastic irrecoverable

Plastic behavior is important for:

Material design for strengthening (alloying)

Work hardening

Hardness – friction and wear

Some materials fracture before yielding (ceramics (brittle))

BUT, fracture can be suppressed by a hydrostatic stress  $\rightarrow$  even ceramics will yield.

Really make sure to understand differences between elastic and plastic states of a system. Understand necking, fracture, and other deformations- understand why these occur.

Good Luck!