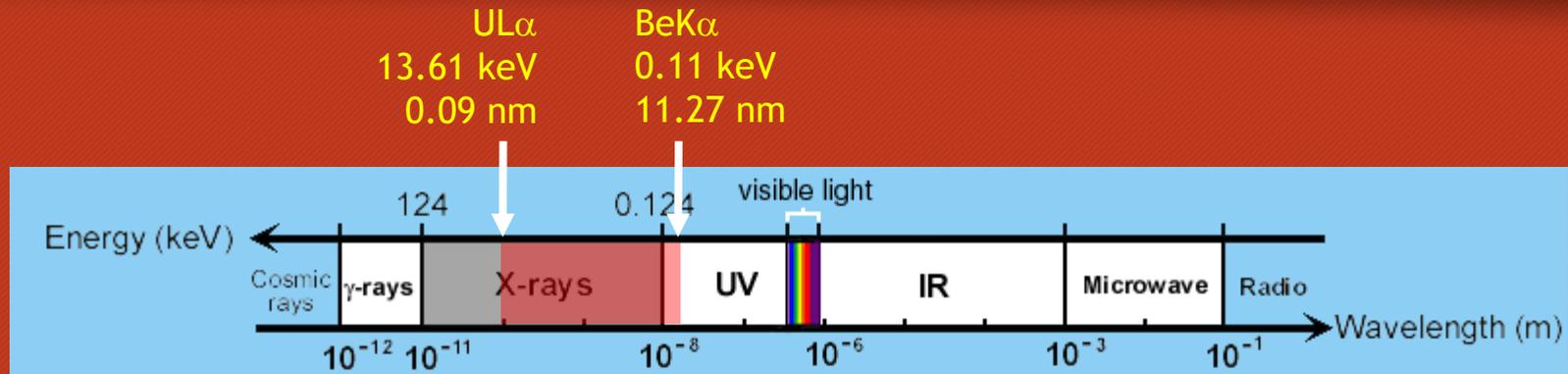


Understanding X-rays: The electromagnetic spectrum

1



$$E = h\nu = h\frac{c}{\lambda} \quad \text{where,} \quad E : \text{energy, } h : \text{Planck's constant, } \nu : \text{frequency}$$

$$c : \text{speed of light in vacuum, } \lambda : \text{wavelength}$$

$$E\lambda = hc = 1.2398 \quad \text{if the units are keV and nm}$$

$$E \text{ (keV)} = h\frac{c}{\lambda} = 1.2398/\lambda \text{ (nm)} \text{ or,} \quad \lambda \text{ (nm)} = h\frac{c}{E} = 1.2398/E \text{ (keV)}$$

$$\text{E.g.,} \quad \lambda_{BeK\alpha} = 11.27 \text{ nm;} \quad \text{Hence, } E_{BeK\alpha} = 1.2398/11.27 = 0.11 \text{ keV}$$

$$E_{UL\alpha} = 13.61 \text{ keV;} \quad \text{Hence, } \lambda_{UL\alpha} = 1.2398/13.61 = 0.09 \text{ nm}$$

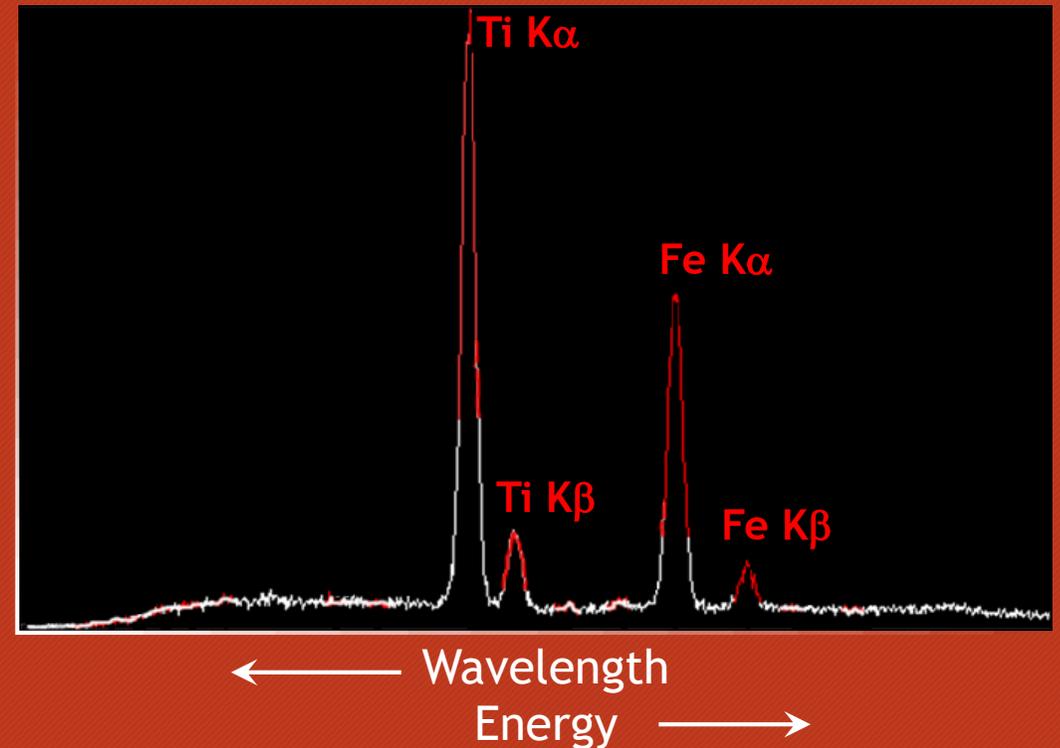
X-ray production in the EPMA

2

X-rays are generated by inelastic scattering of the beam electrons by sample atoms

- *Characteristic X-rays: inner shell interactions*
- *Bremmstrahlung (continuum) X-rays: outer and inner shell interactions*

↑
Intensity

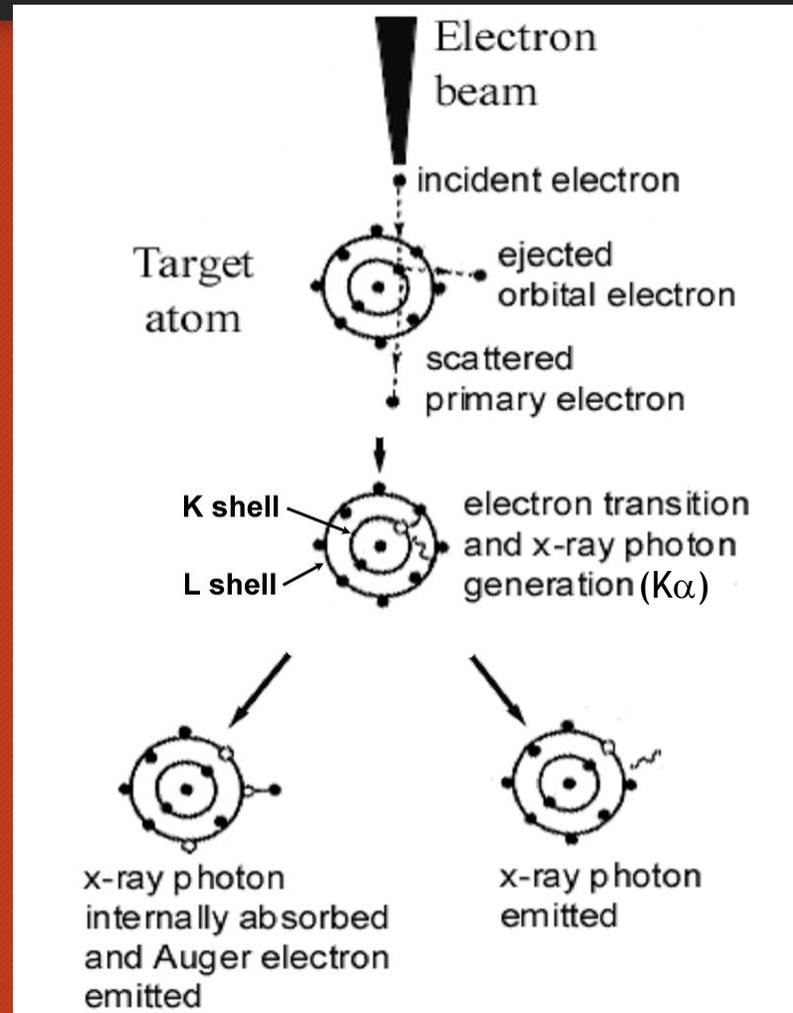


Characteristic X-ray generation

3

*Inner shell ionization
through inelastic scattering*

*followed by electron
transition from an outer
shell to the inner shell*

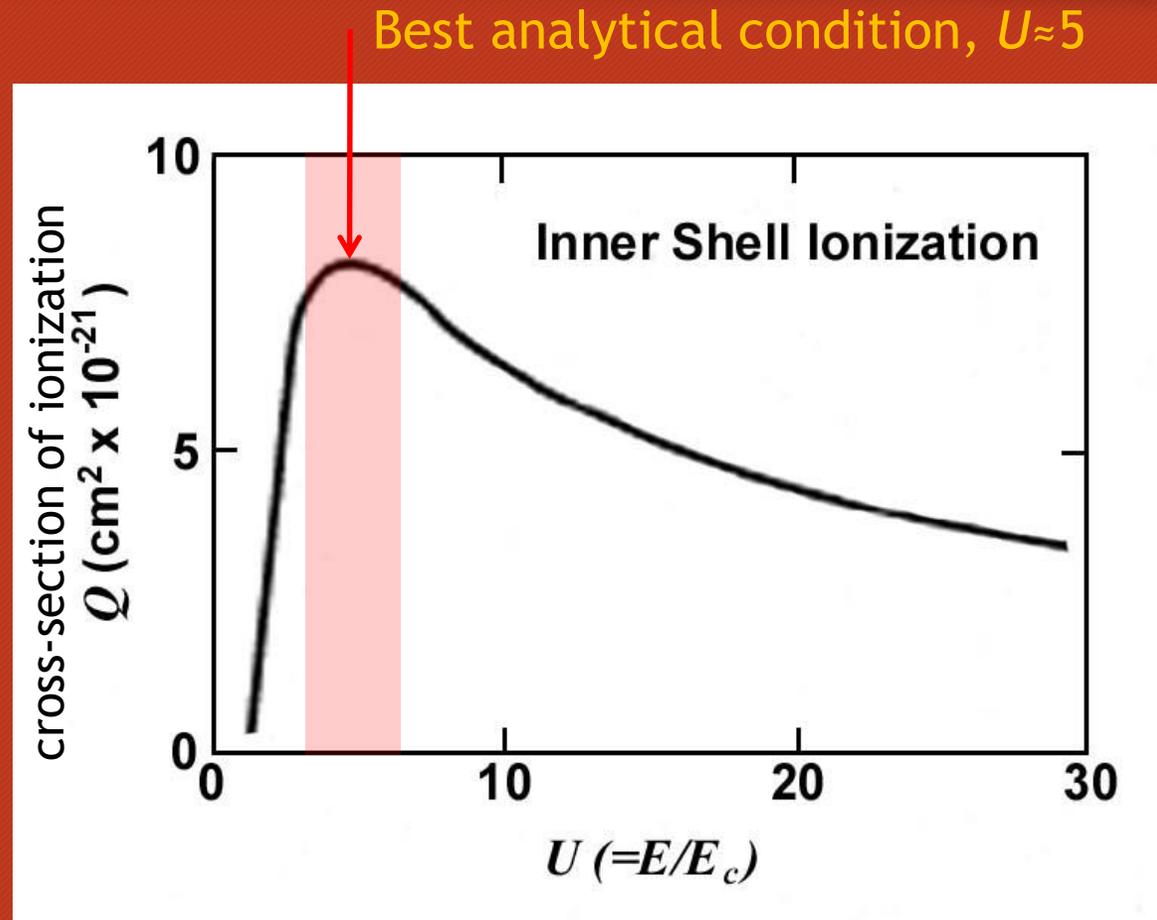


Condition for ionization: Overvoltage

4

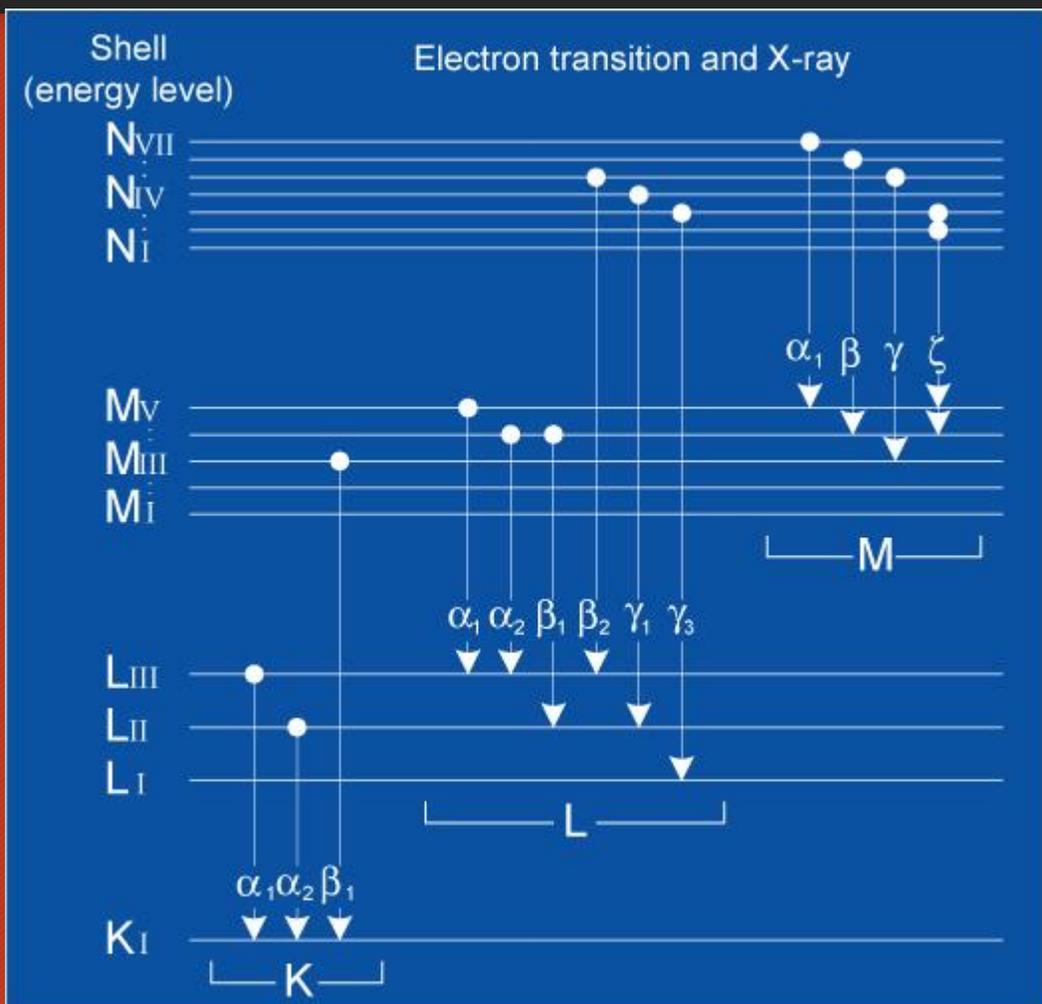
Overvoltage, $U = E/E_c, > 1$

E : electron beam energy
 E_c : critical excitation energy,
ionization energy
of the shell in target atom



X-ray energies

5



X-ray

Electron transition

Energy

Kα

L_{II+III} to K_I

$$E_{K\alpha} = E_{c(K_I)} - E_{c(L_{II+III})}$$

Kβ

M_{III} to K_I

$$E_{K\beta} = E_{c(K_I)} - E_{c(M_{III})}$$

Lα

M_{IV+V} to L_{III}

$$E_{L\alpha} = E_{c(L_{III})} - E_{c(M_{IV+V})}$$

Mα

N_{VII} to M_V

$$E_{M\alpha} = E_{c(M_V)} - E_{c(N_{VII})}$$

Characteristic X-ray energy and critical excitation energy 6

	Uranium		Name
Number	92	U	Symbol
Atomic mass	238.03		
	18.7		
Density (kg/m ³)	Kα 98.434		Characteristic X-ray (keV)
	Lα 13.612		
	M 3.164		

The energy required to generate UK α must be $> E_{c(K)}$ so the overvoltage > 1

$$E_{c(K)} \approx 98.4 + 13.6 + 3.2 \approx 115.2 \text{ keV}$$

Required energy $> 115.2 \text{ keV}$

To calculate $E_{c(K)}$:

Start

$$E_{K\alpha} = E_{c(K)} - E_{c(L)}$$

Rearrange

$$E_{c(K)} = E_{K\alpha} + E_{c(L)}$$

Substitute $E_{c(L)} = E_{L\alpha} + E_{c(M)}$

$$= E_{K\alpha} + (E_{L\alpha} + E_{c(M)})$$

Substitute $E_{c(M)} = E_{M\alpha} + E_{c(N)}$

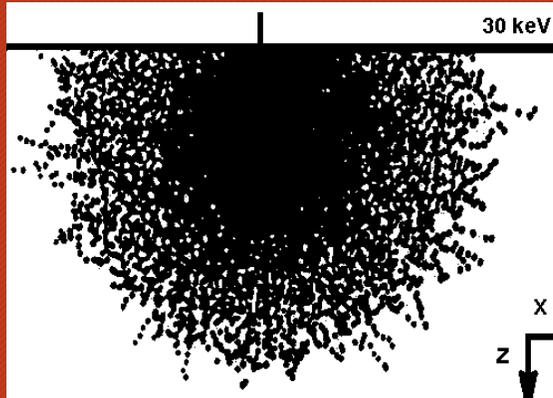
$$= E_{K\alpha} + E_{L\alpha} + (E_{M\alpha} + E_{c(N)})$$

Therefore,

$$E_{c(K)} \approx E_{K\alpha} + E_{L\alpha} + E_{M\alpha}$$

Maximum x-ray production depth (range)

7



$$R_{X\text{-ray}} = 0.033(E^{1.7} - E_c^{1.7}) \frac{A}{\rho Z}$$

(Castaing's formula)

$R_{X\text{-ray}}$ = x-ray range (maximum depth)

E = electron beam energy

E_c = critical excitation energy of target atomic shell

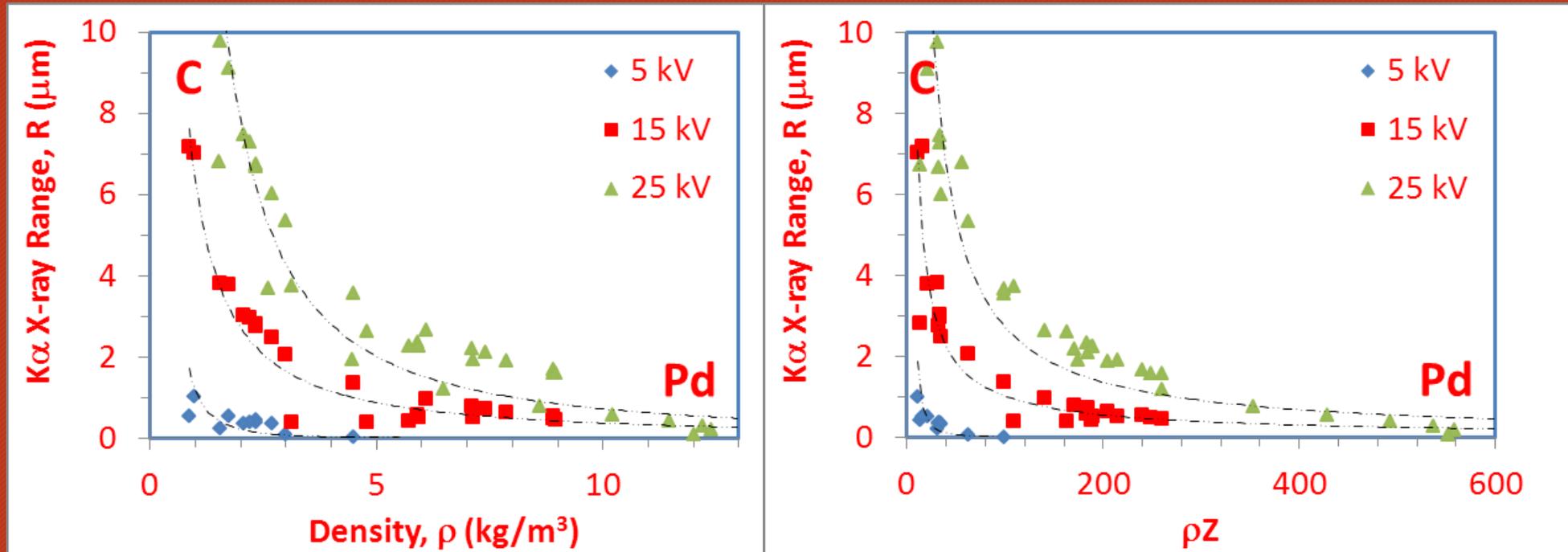
A = atomic weight

ρ = density

Z = atomic number

Maximum x-ray production depth (range)

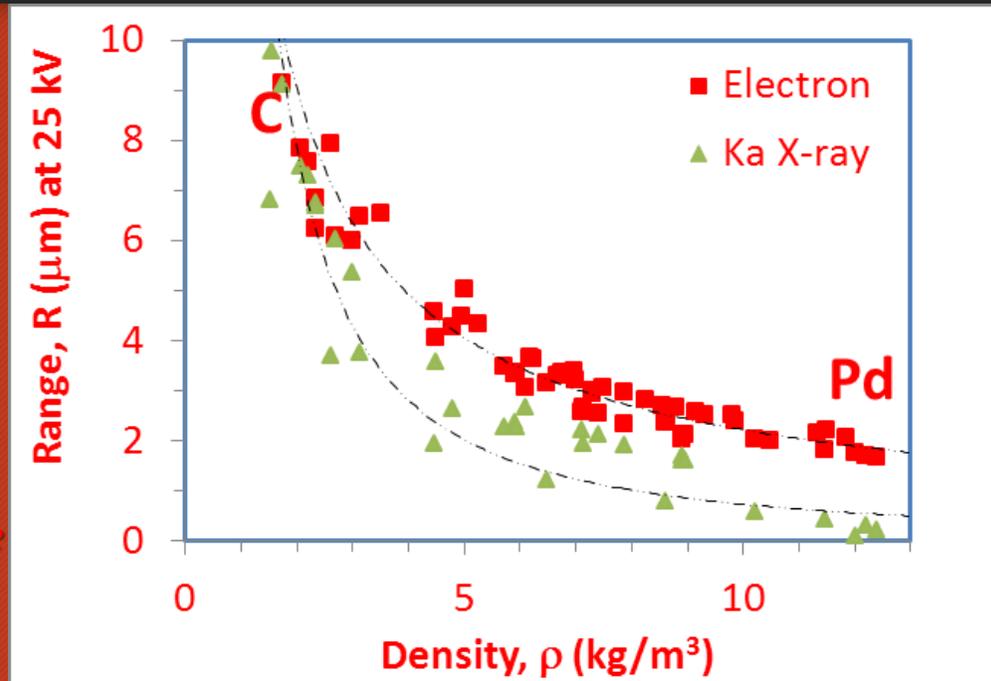
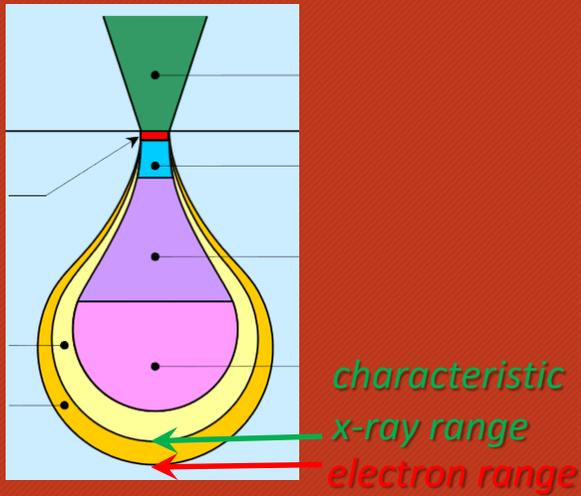
8



Characteristic X-ray range increases with increasing E , and decreasing ρ and ρZ

Electron range versus X-ray range

9



The characteristic x-ray range is always less than the electron range

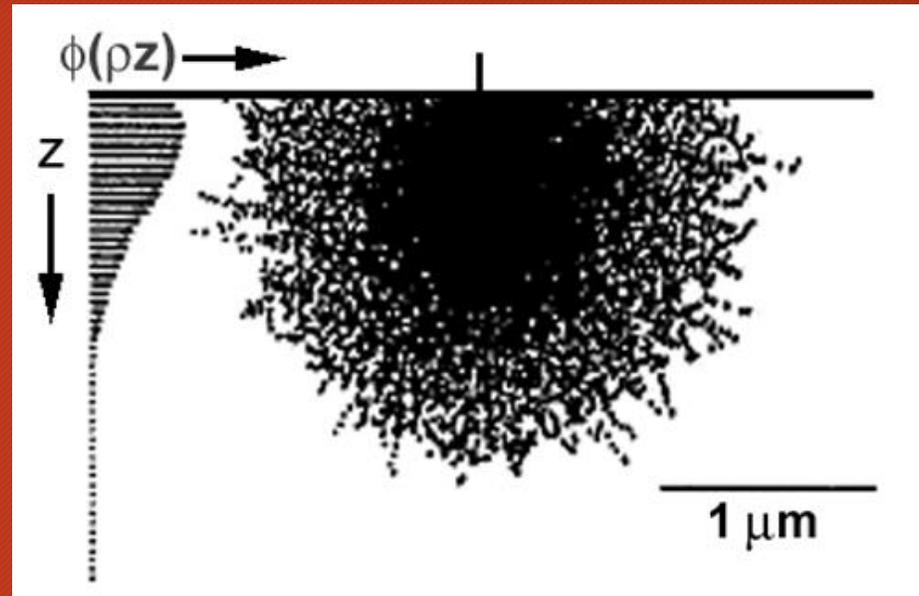
$$R_{\text{electron}} = 0.0276 E^{1.67} \frac{A}{\rho Z^{0.889}}$$

$$R_{\text{x-ray}} = 0.033 (E^{1.7} - E_c^{1.7}) \frac{A}{\rho Z}$$

E = beam energy
 E_c = critical excitation energy of sample atomic shell
 Z = atomic number
 A = atomic weight
 ρ = density

X-ray depth-distribution: the $\phi(\rho z)$ function

10

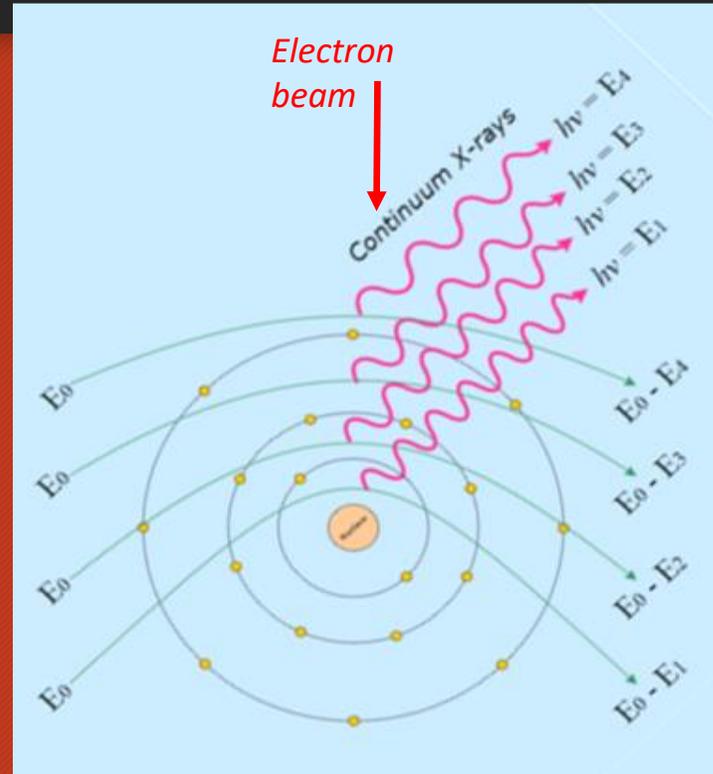


$\phi(\rho z)$ at depth z = intensity from depth 'z' divided by $\phi(\Delta\rho z)$

where, ρ = density, z = depth,
and $\phi(\Delta\rho z)$ = intensity from a free standing layer of thickness ' Δz '

Continuum X-ray generation

11



Produced by deceleration of beam electrons in the electrostatic field of target atoms

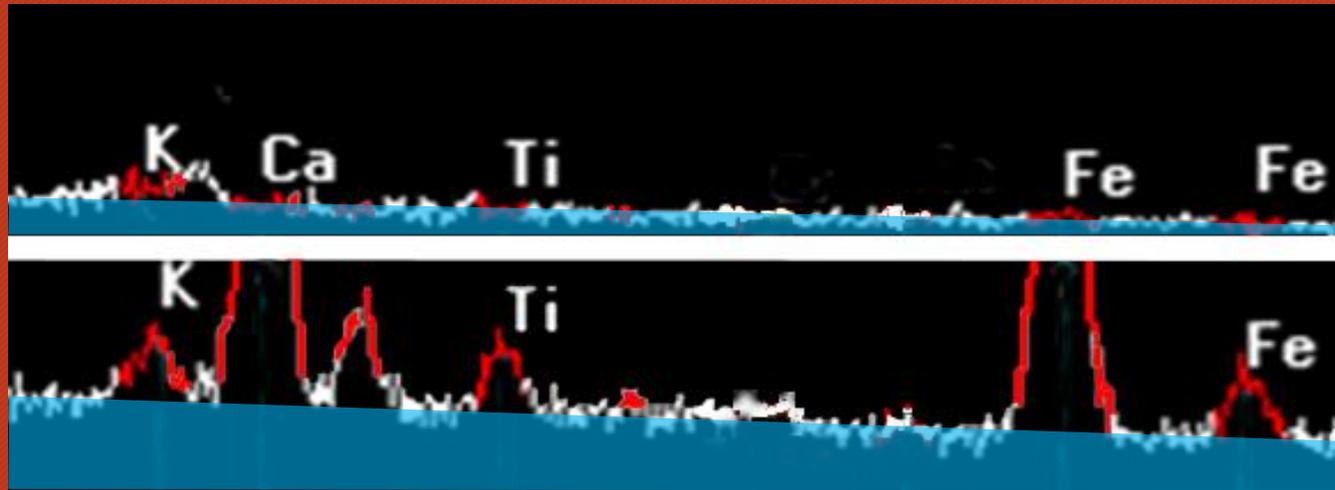
Energy lost by beam electrons is converted to x-ray

(Maximum energy of continuum x-rays = electron beam energy)

Continuum X-rays: background intensity

12

Low-Z sample
(Ca-Fe poor)
Low background

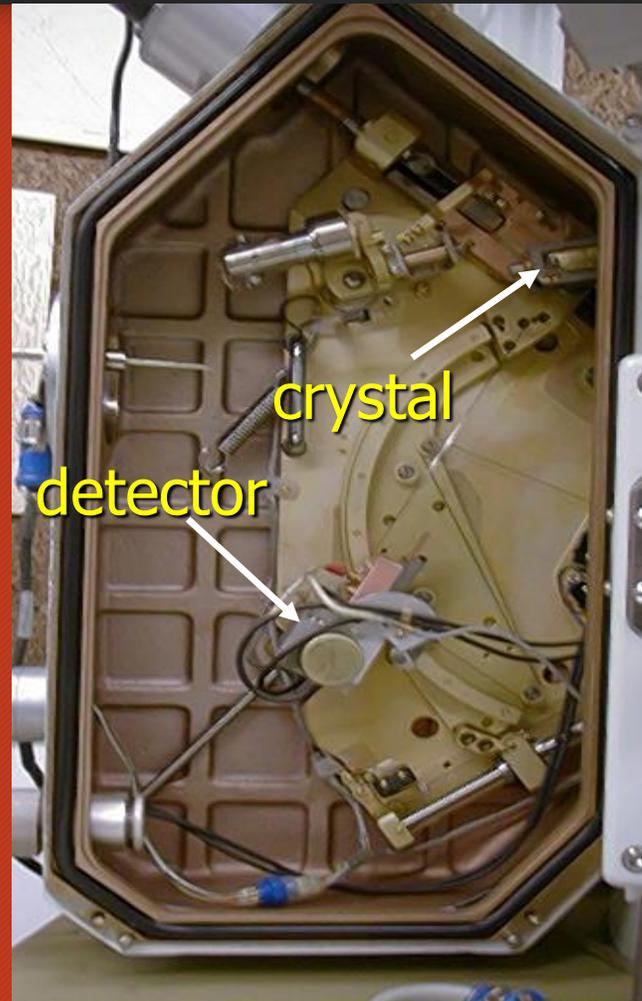


High-Z sample
(Ca-Fe rich)
High background

Increases with sample atomic number

Wavelength Dispersive Spectrometer (WDS)

13

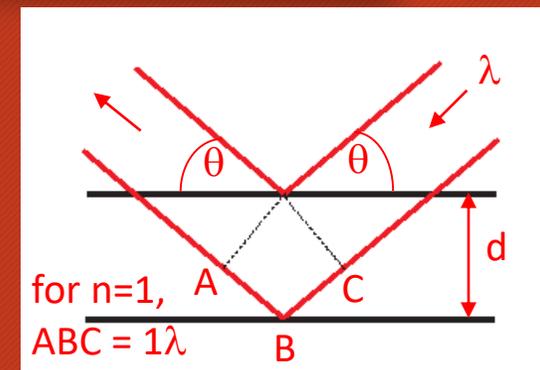
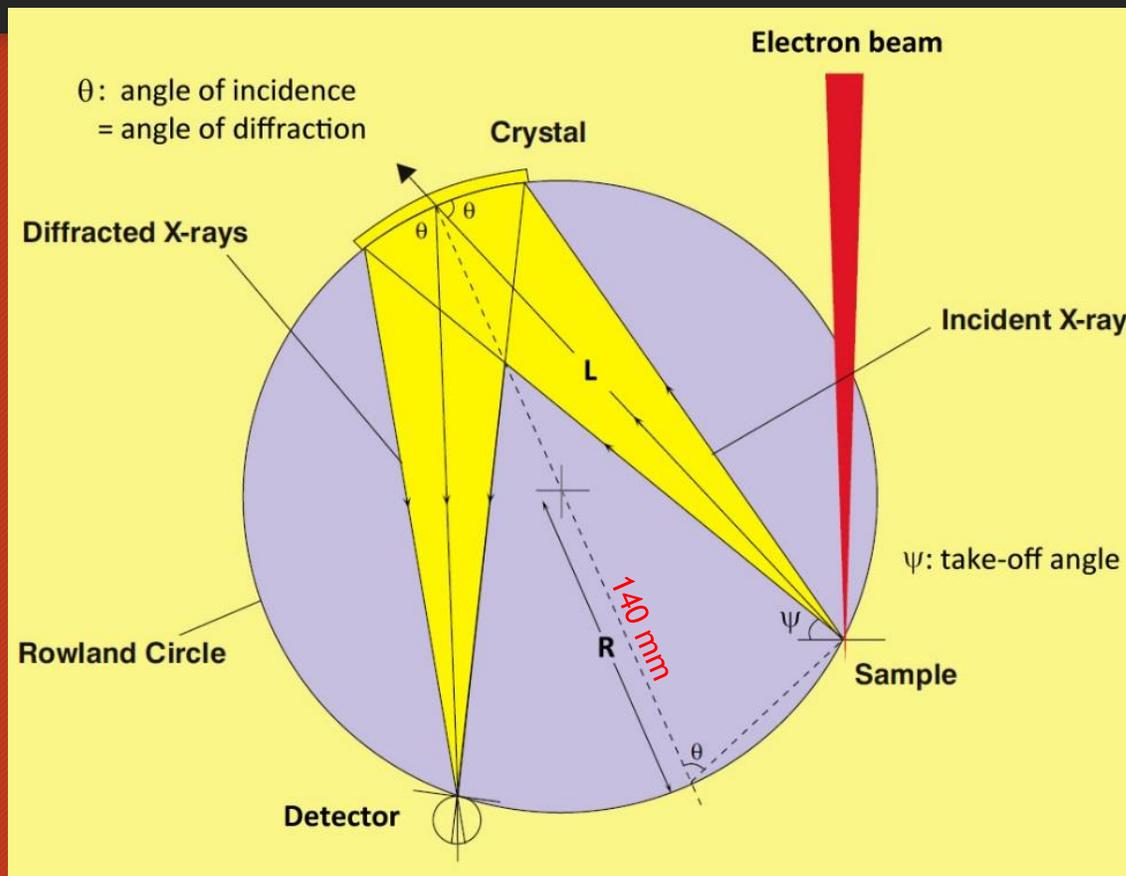


Wavelength Dispersive Spectrometer (WDS)

14

$$\sin \theta = \frac{L}{2R}$$

θ : angle of incidence or diffraction
L: distance between sample and crystal
R: radius of focusing (Rowland) circle



Bragg's Law:

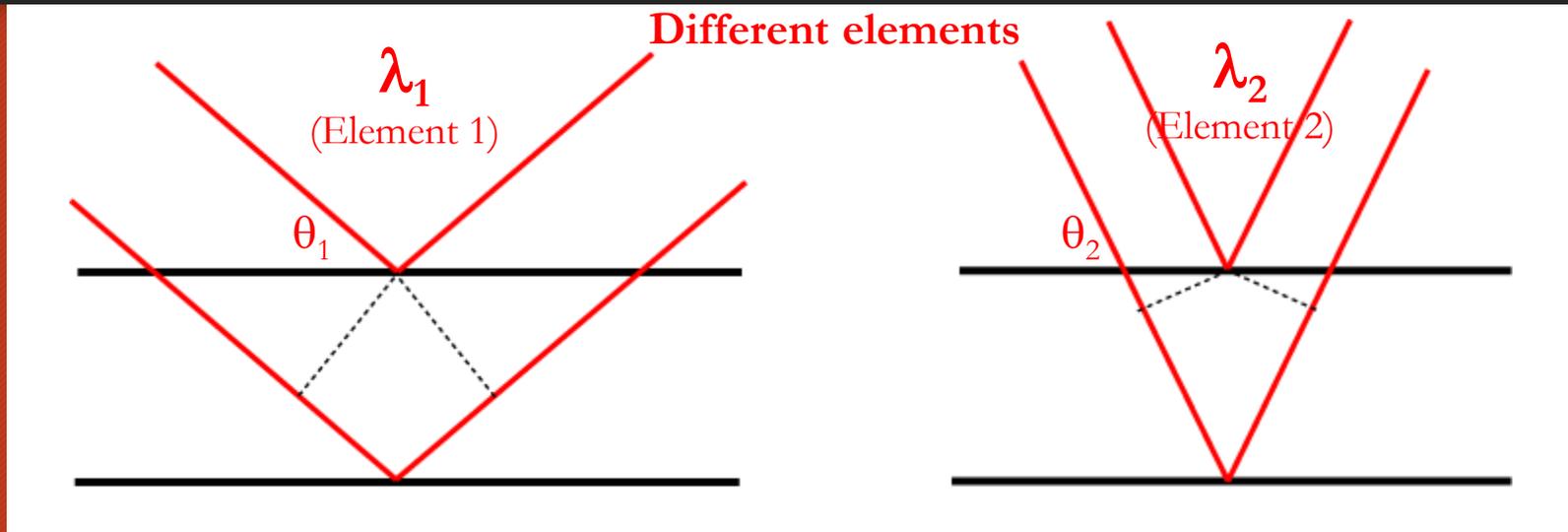
$$n\lambda = 2d \sin \theta$$

n: order of diffraction
 λ : wavelength of X-ray
d: lattice spacing in diffracting crystal
 θ : angle of incidence or diffraction

“L-value”: $L = n\lambda \frac{R}{d}$

Incidence or Diffraction angle

15



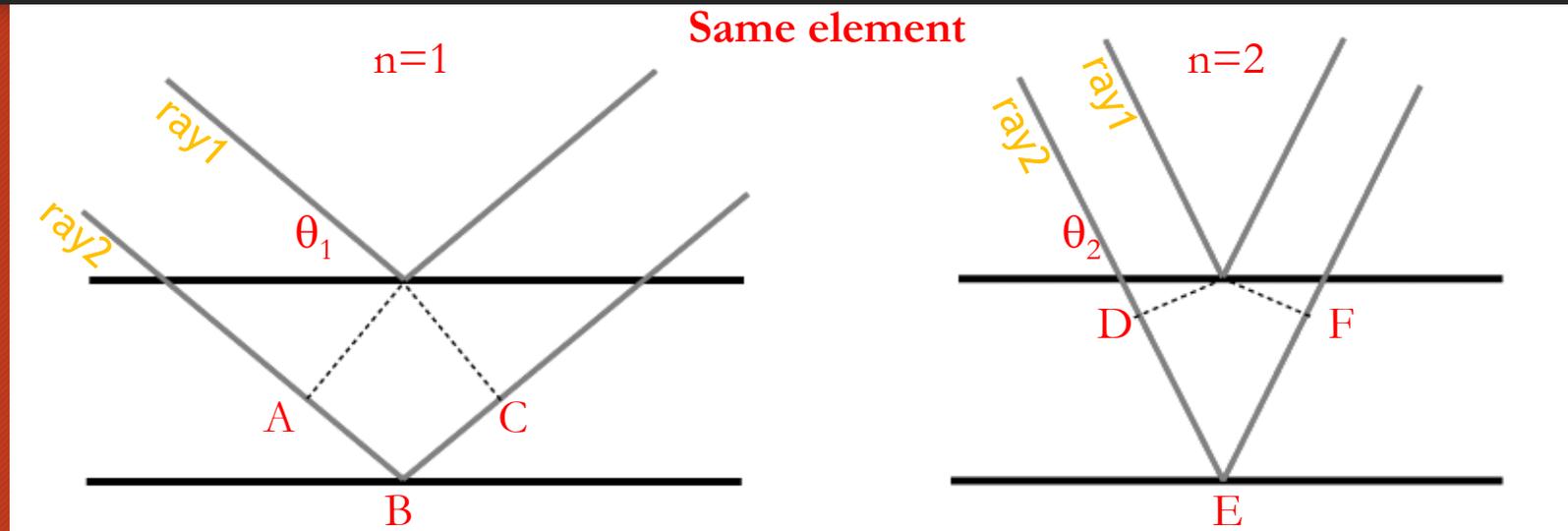
$$n\lambda_1 = 2d \sin\theta_1$$

$$n\lambda_2 = 2d \sin\theta_2$$

*With a different incidence angle, a different wavelength is diffracted
(for the same order of diffraction, n)*

First and second order diffractions

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$$1\lambda = 2d \sin\theta_1$$

$$2\lambda = 2d \sin\theta_2$$

If the incidence angle changes so that $\sin\theta_2 = 2\sin\theta_1$, the 2nd order diffraction of the same wavelength occurs

(path ABC = 1λ ; path DEF = $2 \times$ ABC = 2λ)

In WDS: since $L = 2R \sin\theta$, L-value for the second order diffraction is doubled; $L_2 = 2L_1$

L-value

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Example 1.

Si K α

Energy, E = 1.74 keV

$$\lambda \text{ (nm)} = \frac{1.2398}{E \text{ (keV)}}$$

$$\text{Wavelength, } \lambda = \frac{1.2398}{1.74} = 0.7125 \text{ nm}$$

$$L \text{ (mm)} = n \lambda \text{ (nm)} \frac{R \text{ (mm)}}{d \text{ (nm)}}$$

For n = 1, R = 140, and d_{TAP} = 1.2879,

$$L_{\text{TAP}} = 1 \times 0.7125 \times \frac{140}{1.2879} \\ = 77.45 \text{ mm}$$

Example 2.

U M α

Energy, E = 3.17 keV

$$\lambda \text{ (nm)} = \frac{1.2398}{E \text{ (keV)}}$$

$$\text{Wavelength, } \lambda = \frac{1.2398}{3.17} = 0.3911 \text{ nm}$$

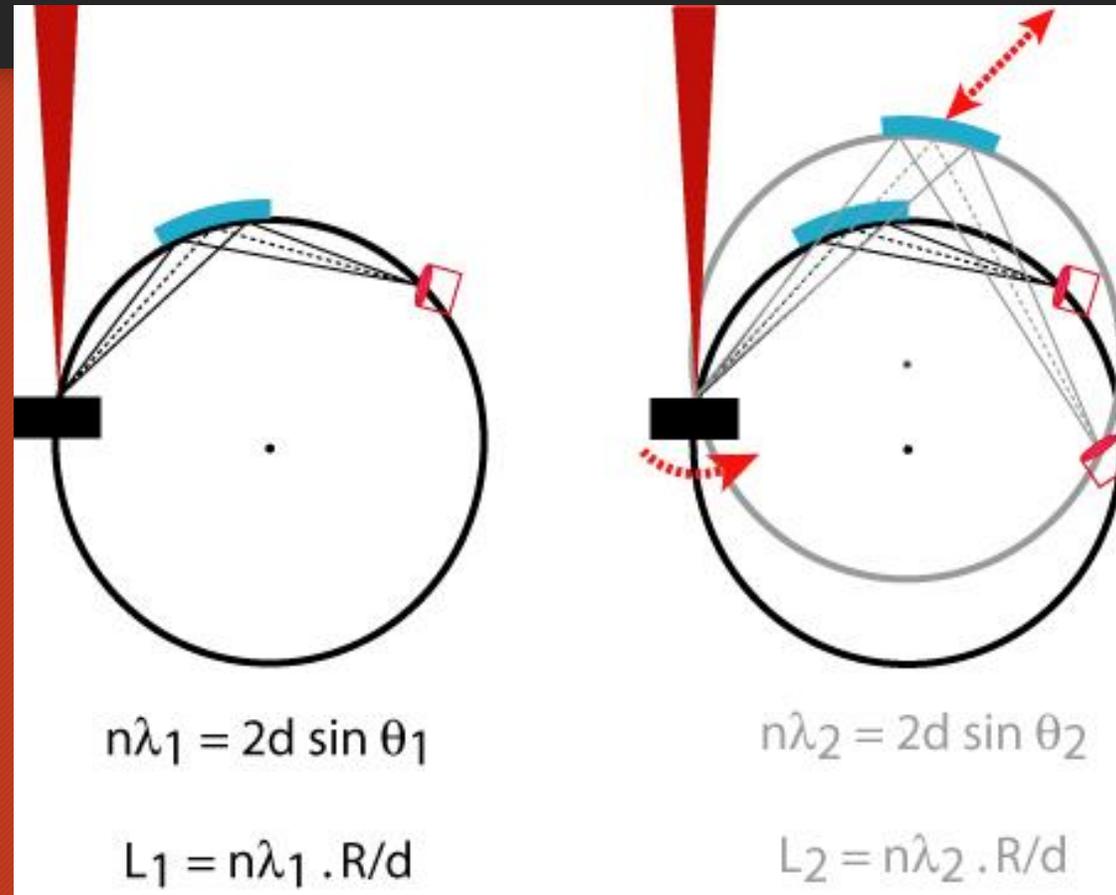
$$L \text{ (mm)} = n \lambda \text{ (nm)} \frac{R \text{ (mm)}}{d \text{ (nm)}}$$

For n = 1, R = 140, and d_{PET} = 0.4371,

$$L_{\text{PET}} = 1 \times 0.3911 \times \frac{140}{0.4371} \\ = 125.27 \text{ mm}$$

WDS operation: changing the L-value

18



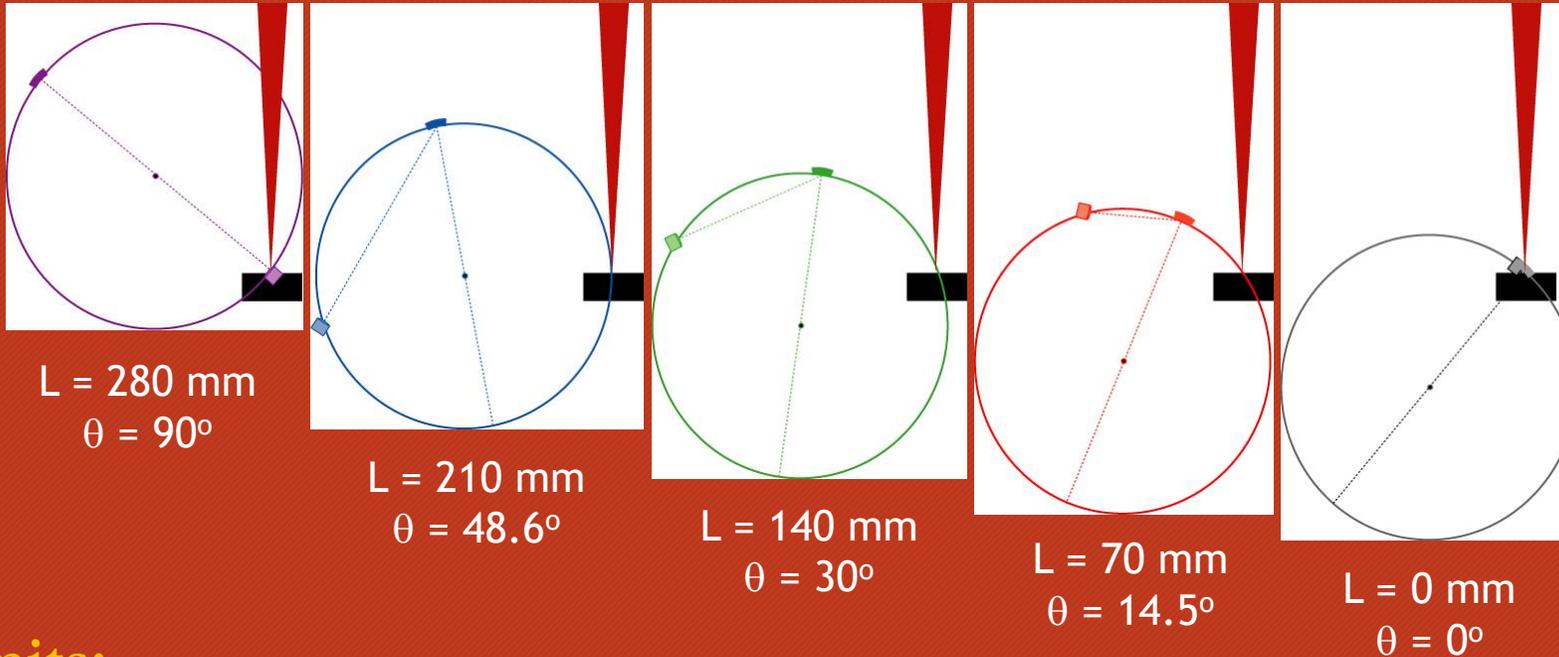
*Radius of focusing circle (R)
remains constant*

*Changing the L-value from L_1 to L_2 changes the incidence angle from θ_1 to θ_2
so that a different wavelength λ_2 is diffracted*

Theoretical limits of spectrometer movement

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For a spectrometer with $R = 140$ mm,



$$L = 2R \sin\theta$$

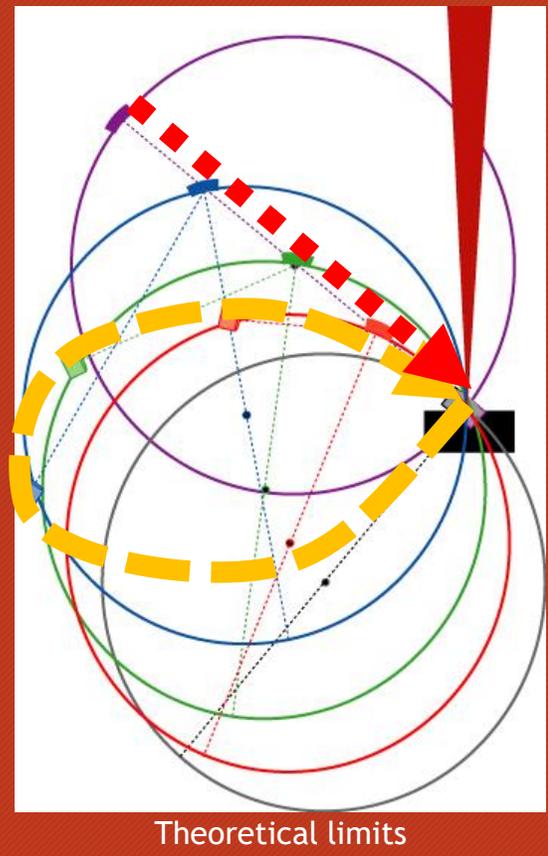
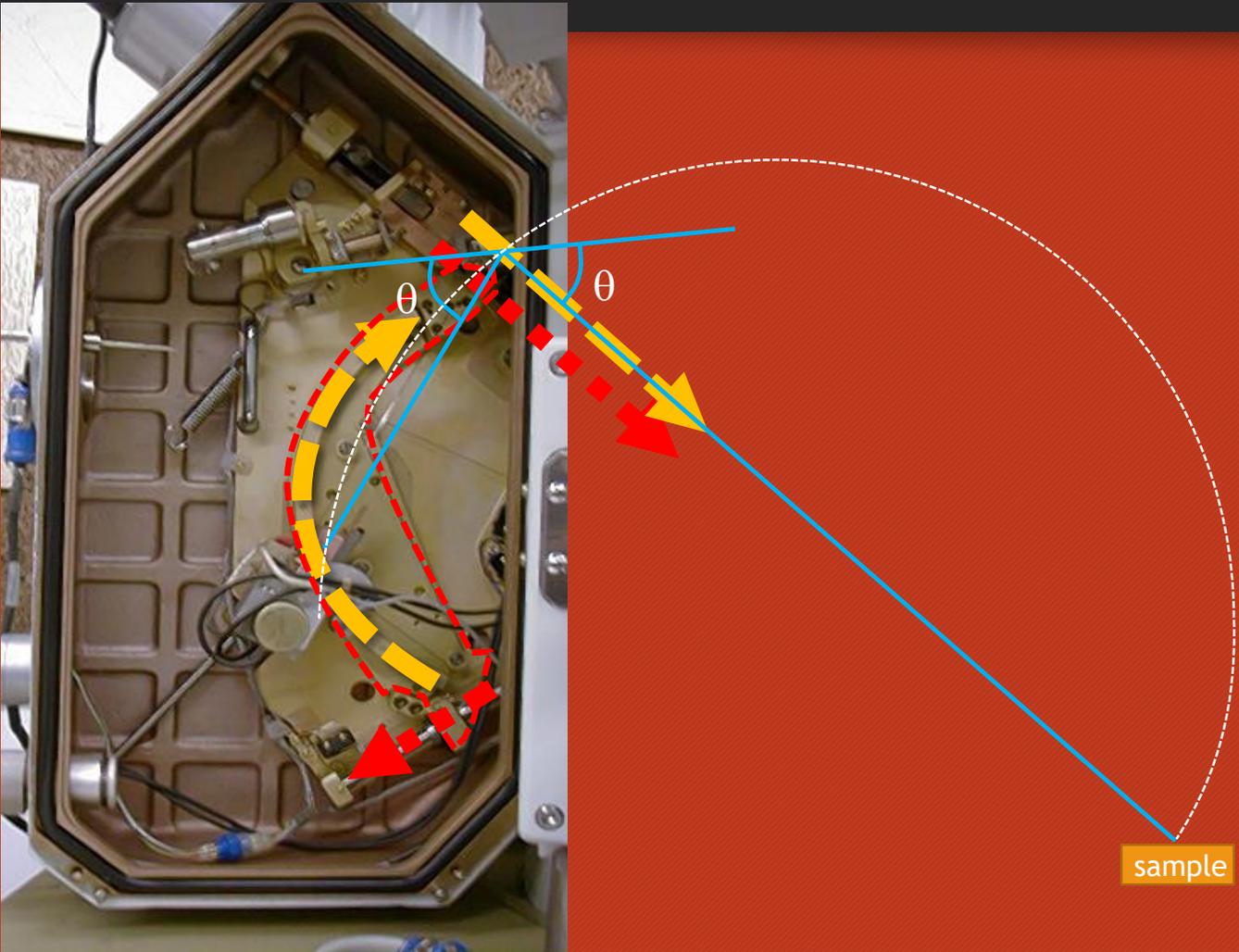
Theoretical limits:

$$2R \geq L \geq 0 \text{ at } 90^\circ \geq \theta \geq 0^\circ$$

$$280 \text{ mm} \geq L \geq 0 \text{ mm}$$

Note θ changes faster between 280-140 mm than between 140-0 mm

Spectrometer movement



Actual limits of spectrometer movement

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For a spectrometer with $R = 140$ mm,

Actual limits: $60 \text{ mm} \leq L \leq 260 \text{ mm}$; $12.4^\circ \leq \theta \leq 68.2^\circ$

Typically, $72.5 \text{ mm} \leq L \leq 229.5 \text{ mm}$; $15^\circ \leq \theta \leq 55^\circ$

Recall $\sin\theta = \frac{L}{2R}$, so $L = 2R\sin\theta$ and $\theta = \sin^{-1}\left(\frac{L}{2R}\right)$

2d of x-ray diffractors

Crystal lattices

	2d (nm)	6 C	14 Si	22 Ti	30 Zn	38 Sr	46 Pd	54 Xe	62 Sm	70 Yb	78 Pt	86 Rn
TAP	2.576	2.36	0.62	2.16	0.57	2.33					0.58	λ (nm)
		⁸ O	¹⁵ P	²⁴ Cr	⁴¹ Nb	⁴⁶ Pd					⁷⁹ Au	L (mm)
		256.5	67.4	234.8	62.0	253.3					63.0	
TAPH	2.576	⁹ F	¹³ Al	²⁴ Cr	³⁵ Br	⁴⁷ Ag				⁷⁰ Yb		
PET	0.8742	¹³ Al	²⁵ Mn	³⁶ Kr		⁶⁵ Tb	⁷⁰ Yb					
PETH	0.8742	¹⁴ Si	²² Ti	³⁷ Rb	⁵⁶ Ba	⁷² Hf						
LIF	0.4027	¹⁹ K	³⁷ Rb	⁴⁸ Cd								
LIFH	0.4027	²⁰ Ca	³¹ Ga	⁵⁰ Sn	⁷⁹ Au							

$K_{\alpha,\beta}$

$L_{\alpha,\beta}$

$M_{\alpha,\beta,\gamma}$

For $n=1$, $\theta = 15$ to 55° ($L = 73$ to 230 mm), and $R = 140$ mm, a crystal can diffract $\sim 0.52d < \lambda < 1.64d$

[recall, $L = n\lambda \frac{R}{d}$, i.e., $\lambda = \frac{L}{nR} d$]

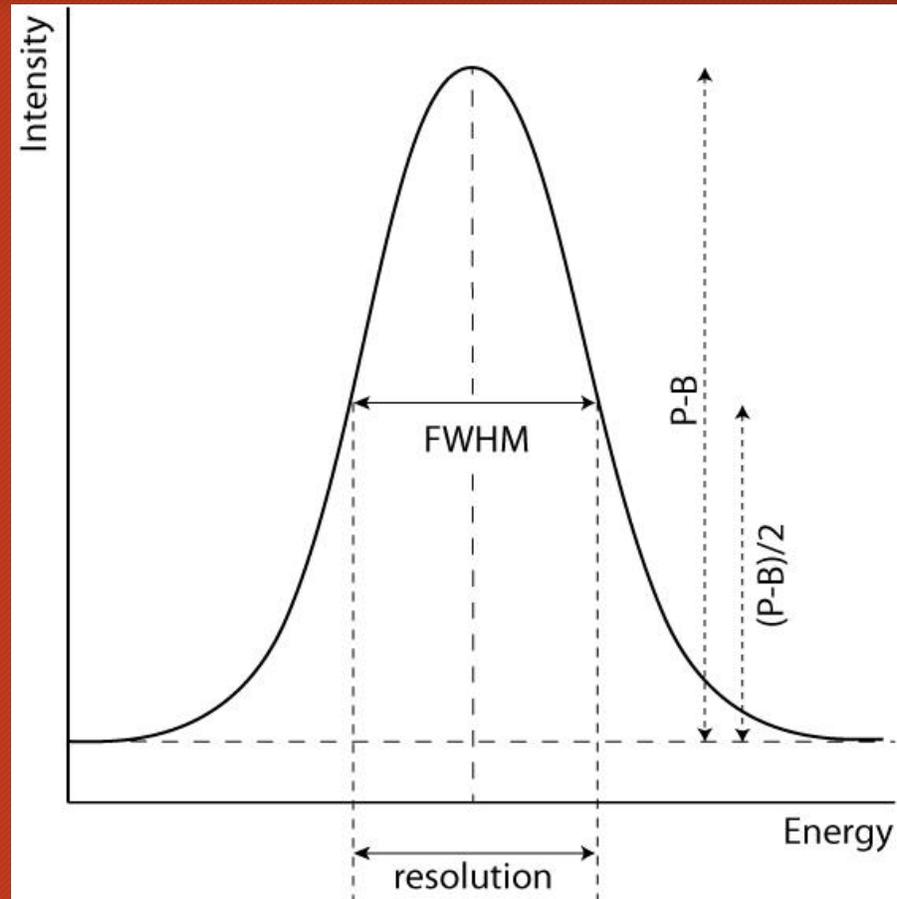
Layered structures

	2d (nm)	Be	B	C	N	O	F
NSTE	Approx.10		○	○	○	○	
LDE1	Approx.6			△	◎	◎	◎
LDE2	Approx.10		◎	◎	◎	○	
LDEB	Approx.14.5	◎	○				
LDE1H	Approx.6			△	◎	◎	
LDE2H	Approx.10		◎	◎			
LDENH	Approx.8			○	◎		
LDE3H	Approx.20	◎	○				
LDE5H	Approx.8			◎	◎		
LDEBH	Approx.14.5	◎	◎				

$\lambda_{BeK\alpha} = 11.27$ nm; so $BeK\alpha$ can be diffracted only by diffractors with $2d > 11.27$ nm
e.g., with LDE3H (at $L = 157.8$ mm), and LDEB and LDEBH (at $L = 217.6$ mm)

Spectral resolution

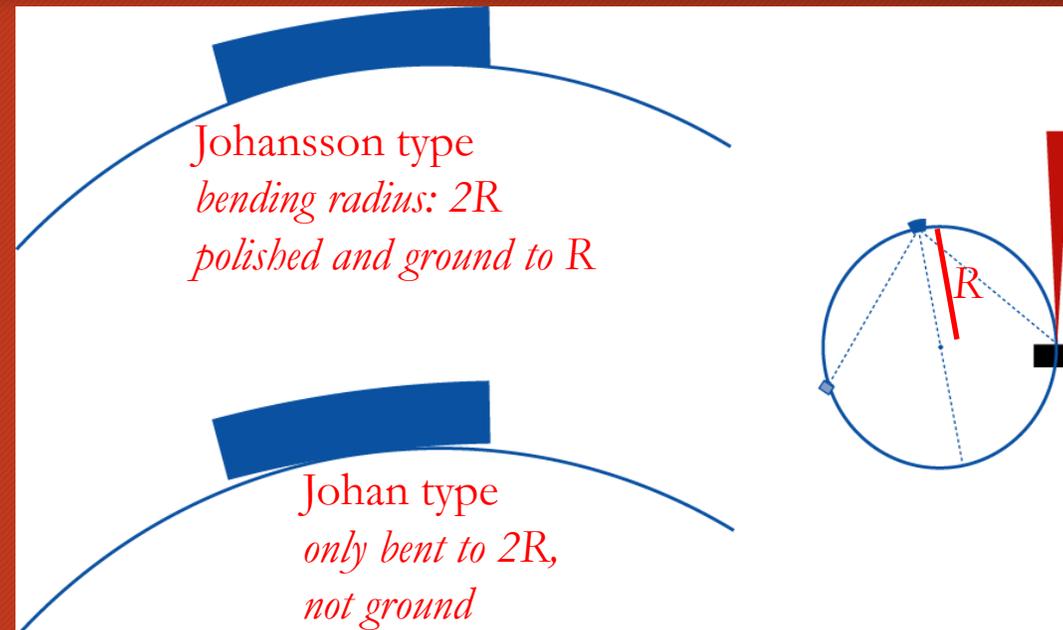
23



Full-Width Half-Maximum (FWHM)

Curved diffracting crystals

24

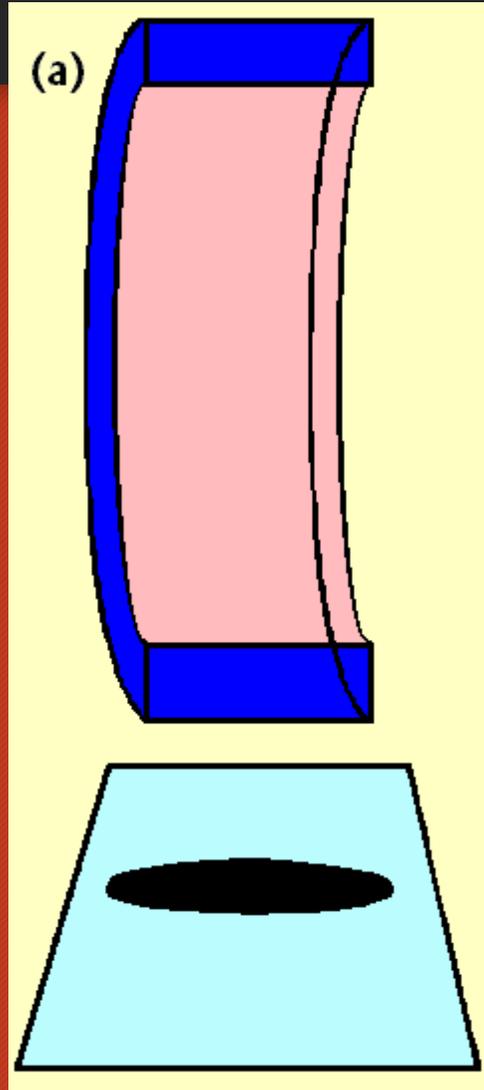


Peak resolution with fully focusing **Johansson-type** crystal: FWHM ~ 10 eV

Some defocusing in **Johan-type**, but resolution is not compromised

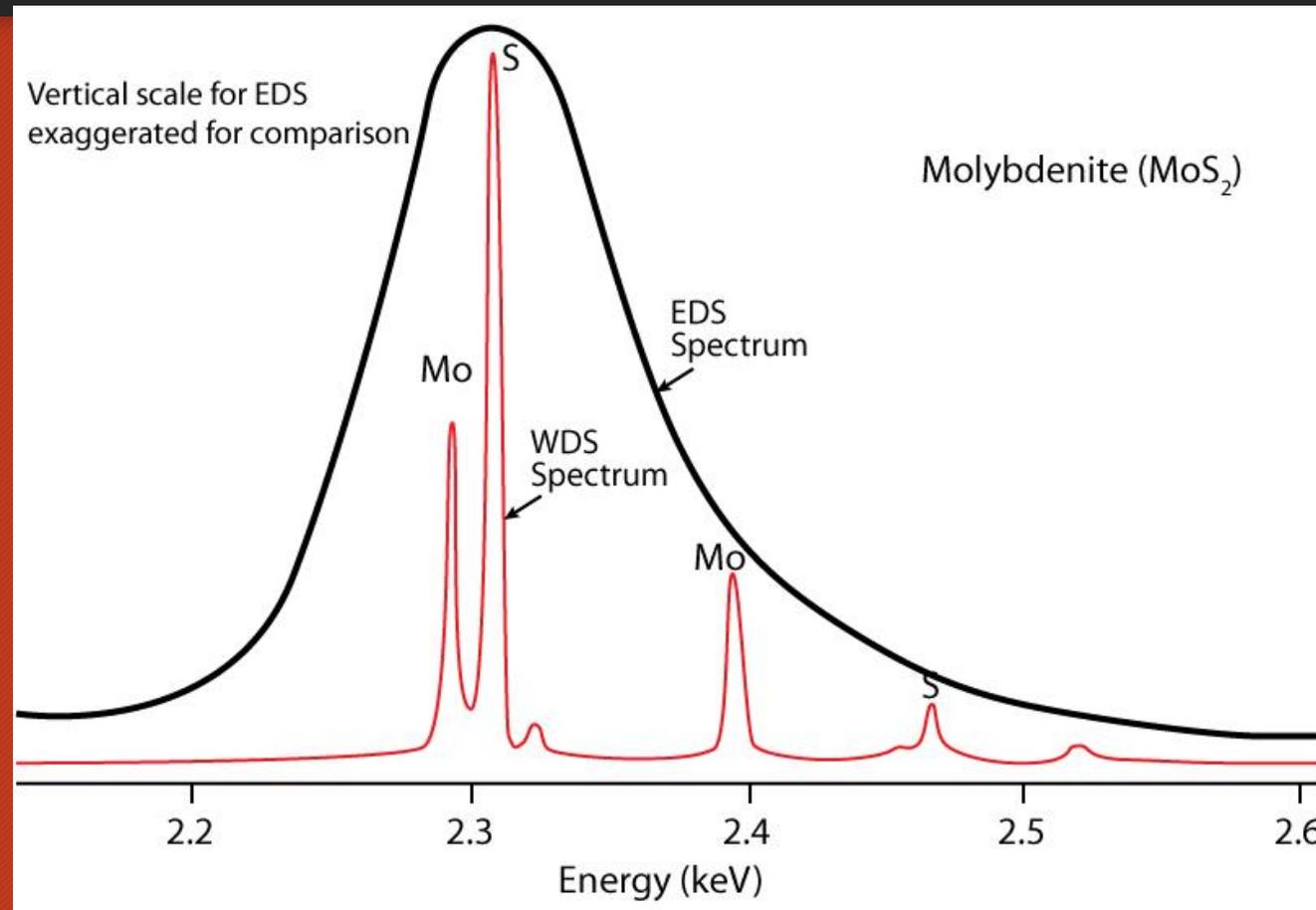
X-ray focusing ellipsoid

25



WDS vs. EDS spectral resolution

26



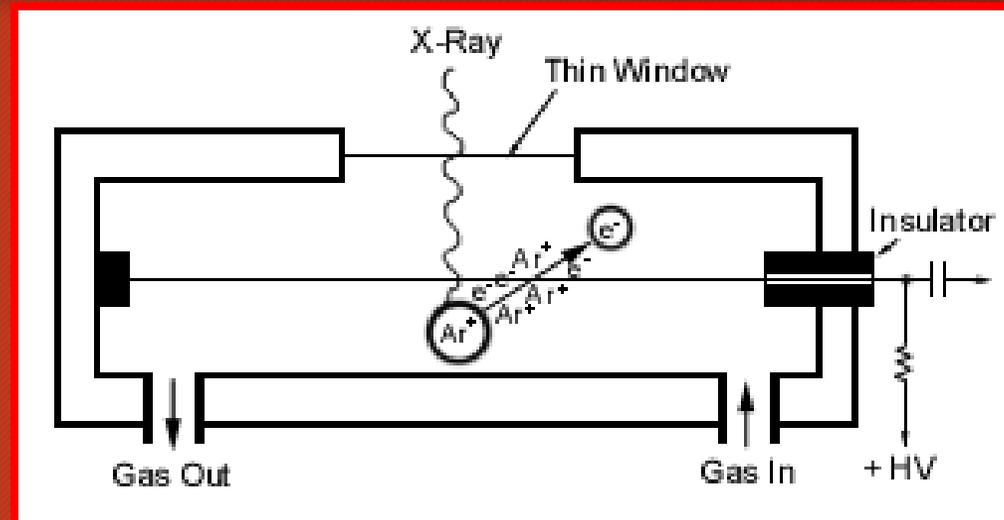
Peak overlaps in EDS spectrum

Peak resolution with WDS (FWHM ~10 eV) is an order of magnitude better than with EDS (FWHM ~150 eV)

WDS detector: Proportional counter

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*Tungsten collection wire
at 1-3 kV voltage
Normal operation:
1600-1850 V*

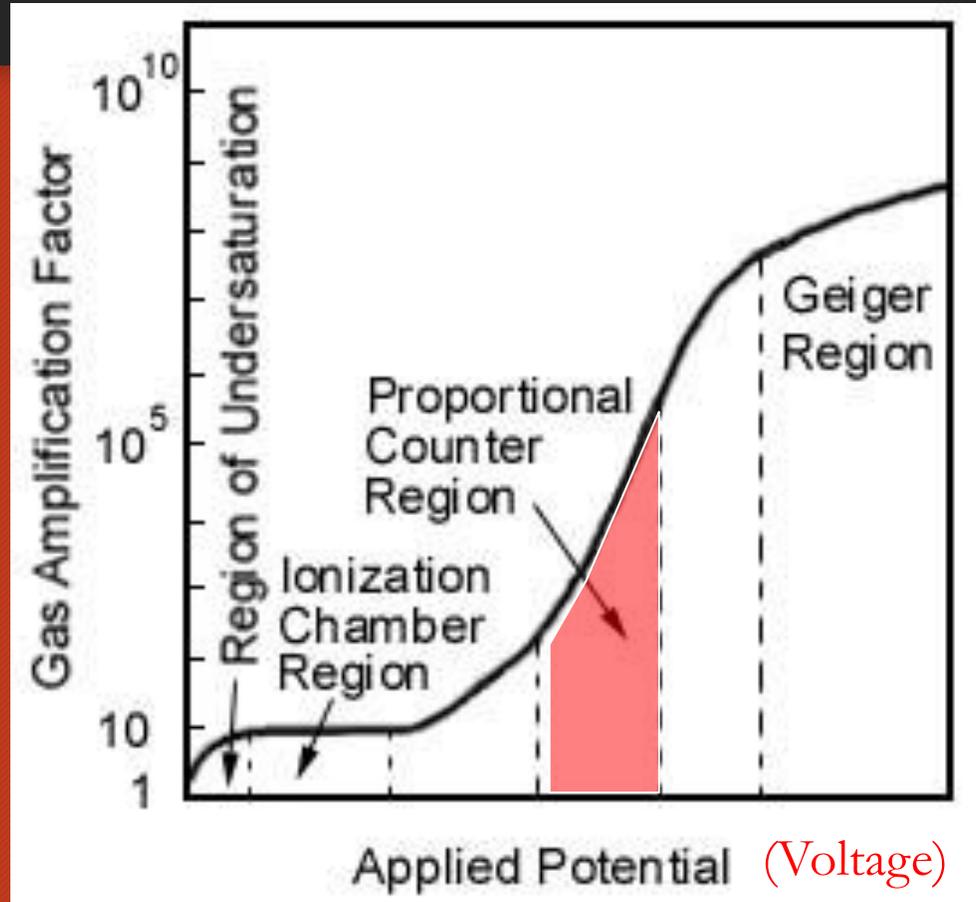


- *Incoming x-ray ionizes a gas atom that sets up a **chain of ionizations in the gas**. The signal is thus amplified by the gas itself.*
- ***Pulse voltage** generated is **proportional to the voltage** in the collection wire under normal operating conditions.*

- *Flow counter:*
 - *P-10 gas (90% Argon + 10% methane quenching agent)*
 - *Polypropylene window*
- *Sealed counter:*
 - *Xenon gas*
 - *Beryllium window*

Signal amplification

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The amplification factor is proportional to the voltage in the collection wire in the proportional counter region

Typical voltage range in the proportional counter region for a W wire: 1600-1850 V

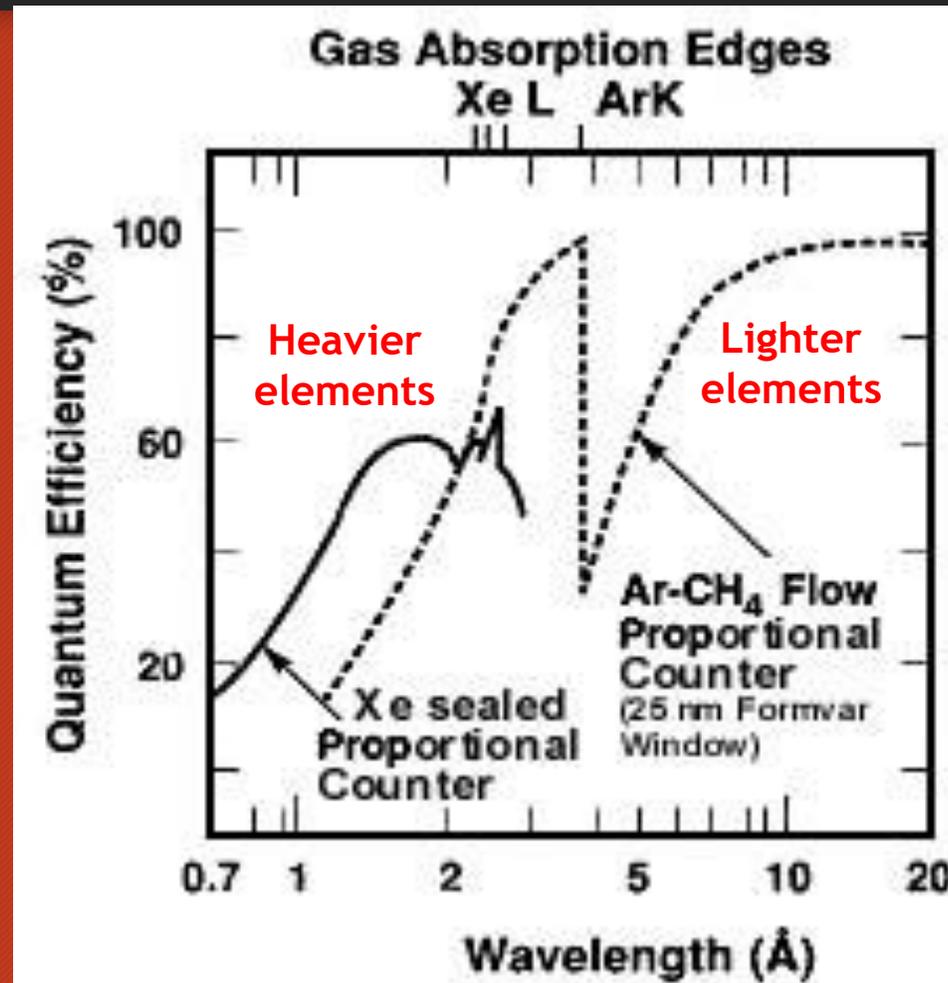
Quantum efficiency of counter gas

29

Highest when the incoming X-ray is least absorbed by the gas

Decreases when the X-ray is absorbed by ionizing an inner shell of the gas atom, generating $\text{ArK}\alpha$ or $\text{XeL}\alpha$

Lowest when $E_{\text{X-ray}}$ is slightly higher than the $E_{\text{c(Ar K-shell)}}$ or $E_{\text{c(Xe L-shell)}}$ absorption edges

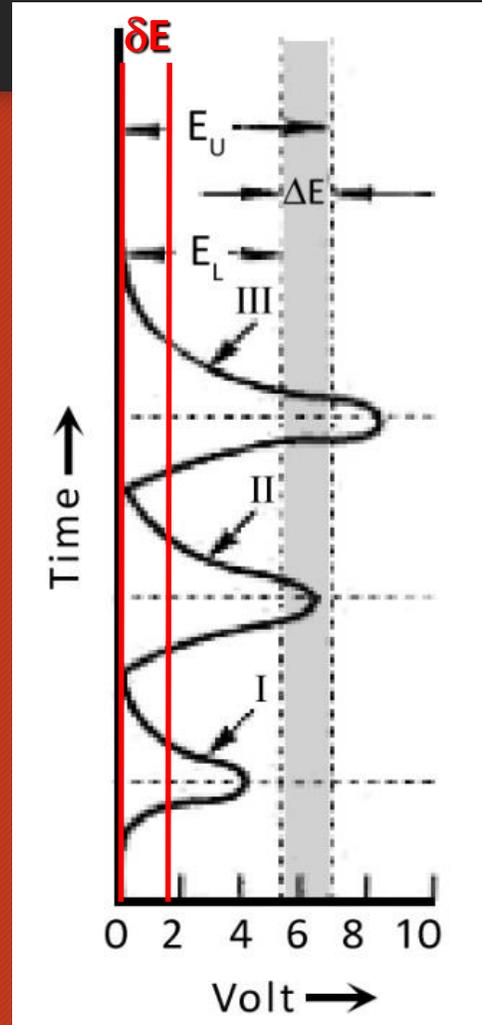


- Argon: *long wavelength (low energy) detection*
- Xenon: *short wavelength (high energy) detection*

Proportional counter setup: Pulse Height Analysis

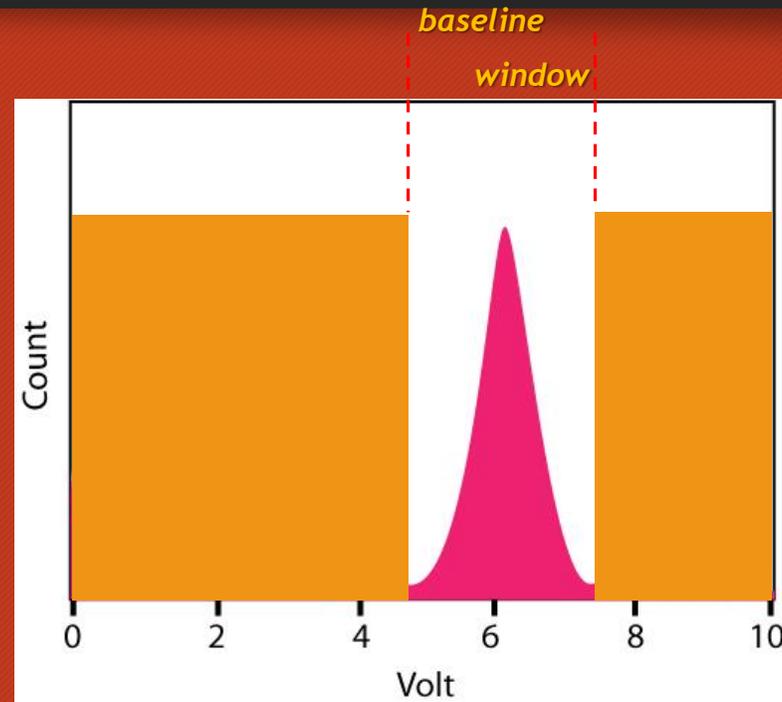
30

Proportional counter output:
Voltage pulses from noise
and x-ray signal



A **Single Channel Analyzer (SCA)** can be set to allow only x-ray voltage pulses within ΔE to pass through

ΔE is determined by **Pulse Height Analysis (PHA)** through an SCA scan

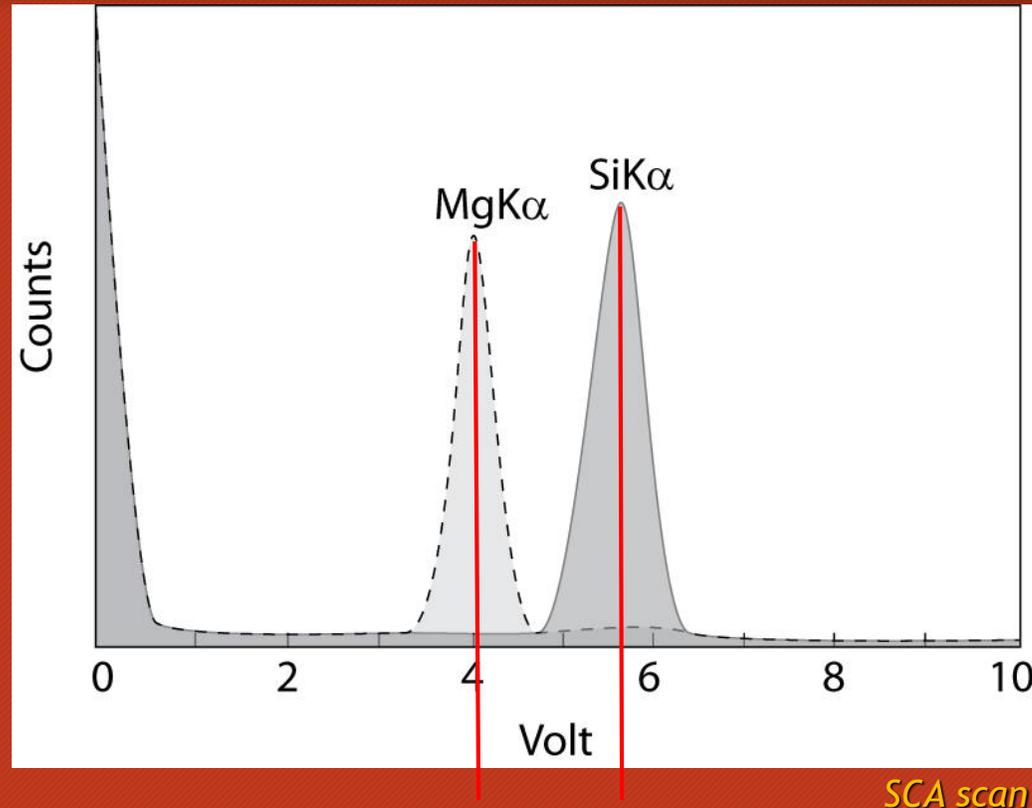


An SCA scan shows the variation in count rate as a small voltage window (δE) is moved across the voltage range

Baseline and window voltages (ΔE) are set to filter out noise and unwanted signal

Pulse voltage in SCA scan

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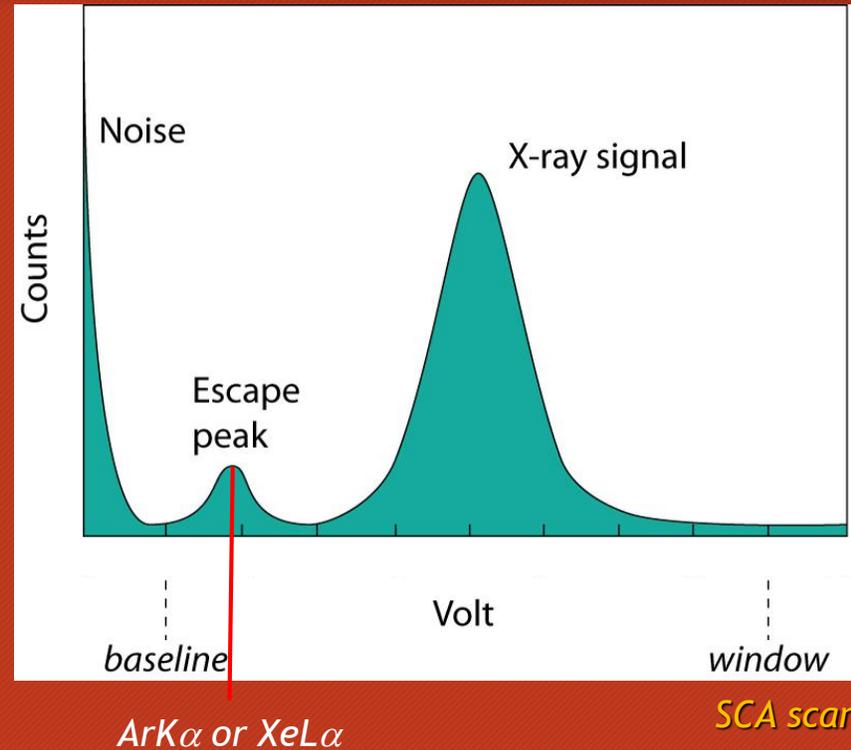
Pulse voltage is proportional to energy of the X-ray being detected

Energy of **SiK α** (1.739 keV) is **~1.39** times the energy of **MgK α** (1.253 keV)

If the pulse for **MgK α** is at 4 V, the pulse for **SiK α** will be at $4 \times 1.39 = 5.56$ V

Escape peak in SCA scan

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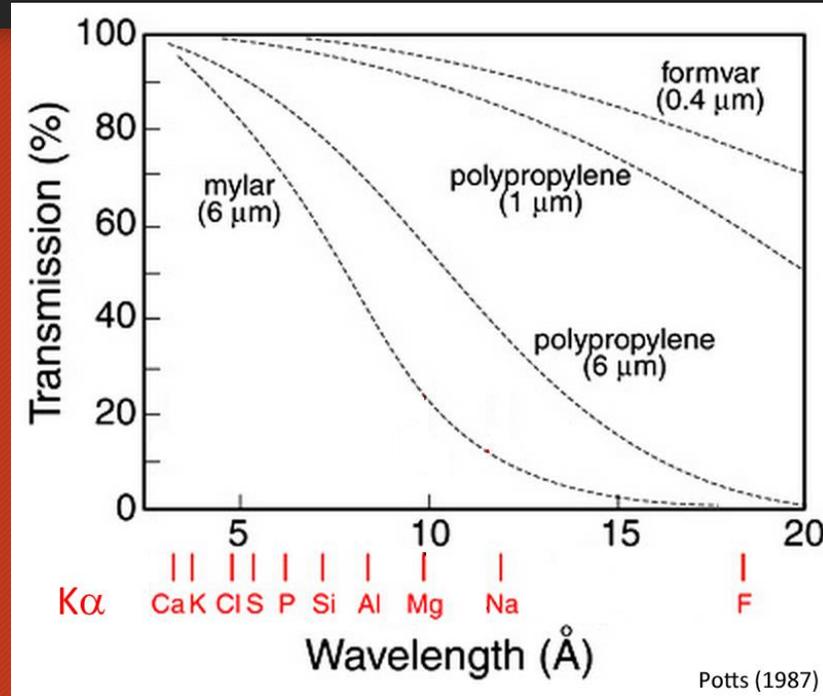
Escape peaks fluoresced by incoming X-ray:

- P-10 counter: ArK α
- Xenon counter: XeL α

If the pulse for NiK α (7.47 keV) is at 5.20 V,
the XeL α (4.11 keV) escape peak will be at $5.2 \times [(7.47 - 4.11) / 7.47] = 2.34$ V

Proportional counter window material

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- Mylar has lower transmittance than polypropylene, especially for light element x-rays
- Thin windows are better for light elements

1 μm thick polypropylene window transmits ~60% of the F Kα

6 μm thick polypropylene window transmits only ~5% of the F Kα

Detector slit

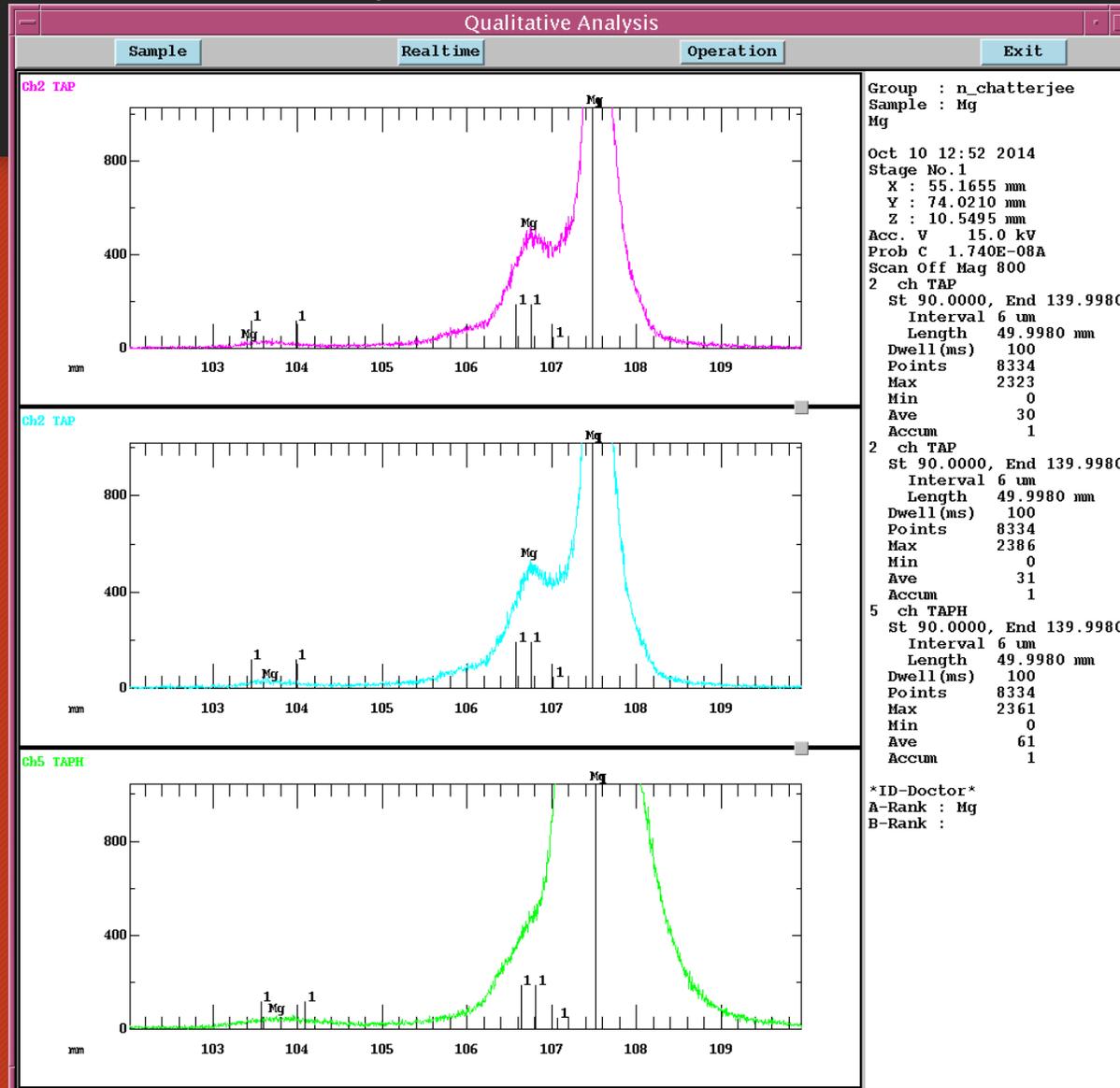
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- *Positioned in front of the proportional counter window*
- *Cuts off stray x-rays and electrons*

<i>Open:</i>	<i>LDE</i>	<i>P-10 flow counter</i>	<i>Very light elements (very low E, very long λ)</i>
<i>550-300 μm:</i>	<i>PET or LIF</i>	<i>Xe sealed counter</i>	<i>Heavy elements (high E, short λ)</i>
<i>300 μm:</i>	<i>TAP</i>	<i>P-10 flow counter</i>	<i>Light elements (low E, long λ)</i>
<i>300 μm with Mylar film:</i>	<i>PET or LIF</i>	<i>P-10 flow counter</i>	<i>Heavy elements (high E, short λ)</i>

Semi-quantitative analysis

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Compositional imaging with X-rays: elemental mapping

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- **Beam-rastered image:**

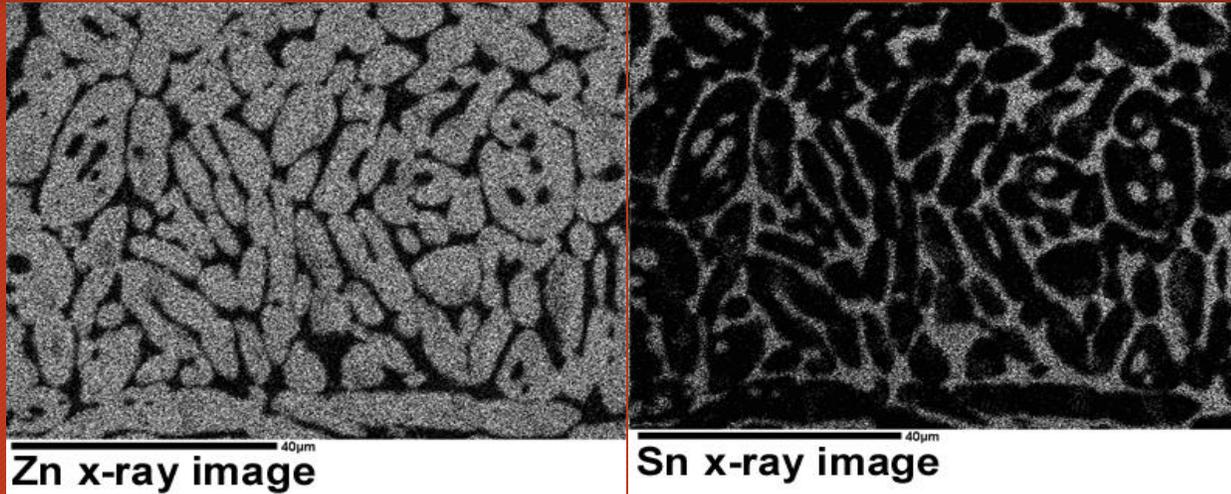
electron beam rasters over the area to be imaged

- **Stage-rastered image:**

electron beam is stationary, stage moves

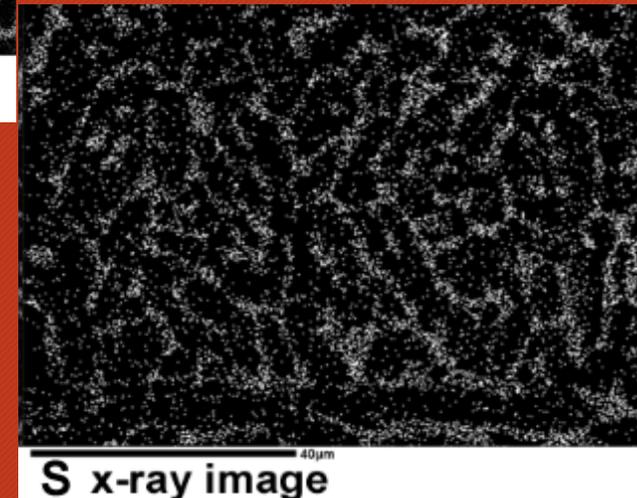
Background in x-ray image

37

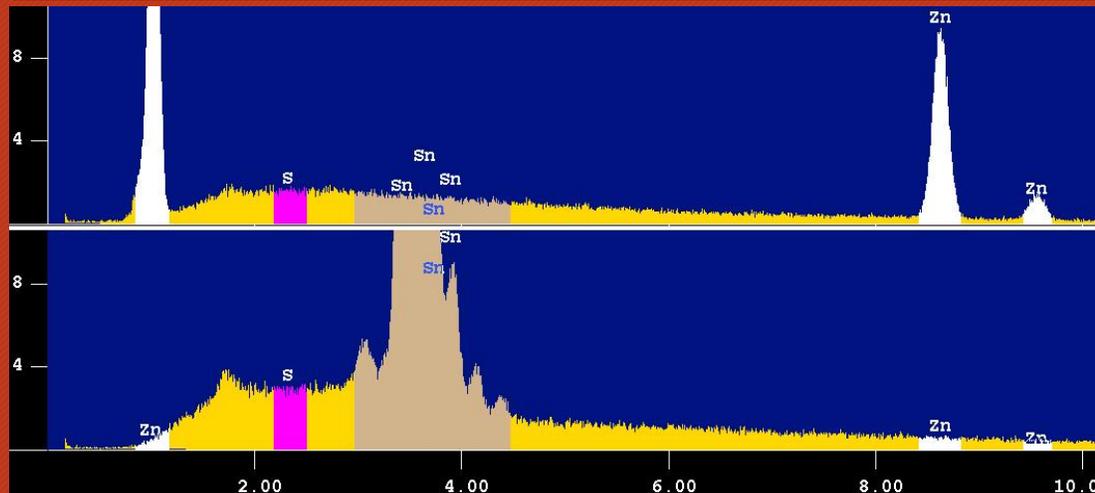


Zn-Sn composite

Background image



Zn-rich phase
(low Z)



Sn-rich phase
(high Z)

X-ray defocusing in beam-rastered image

38

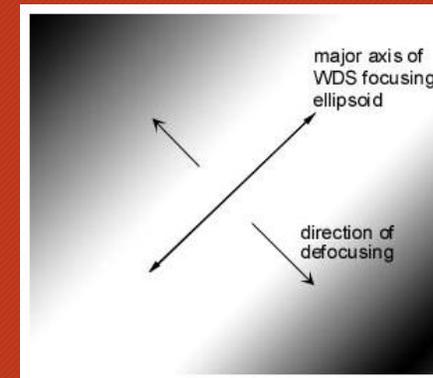
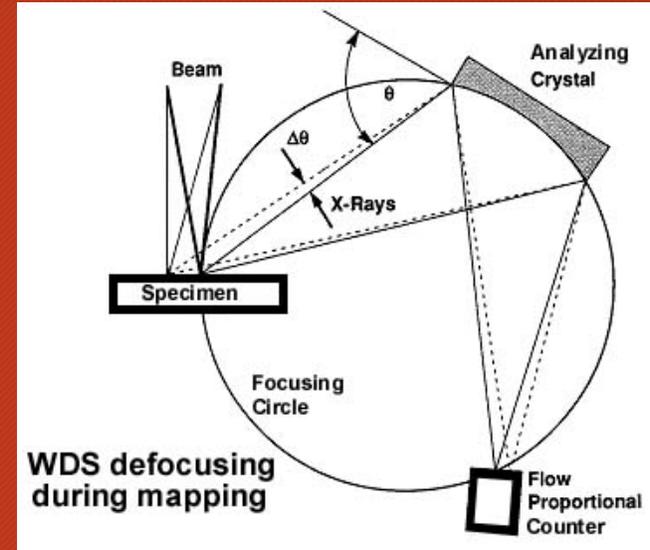
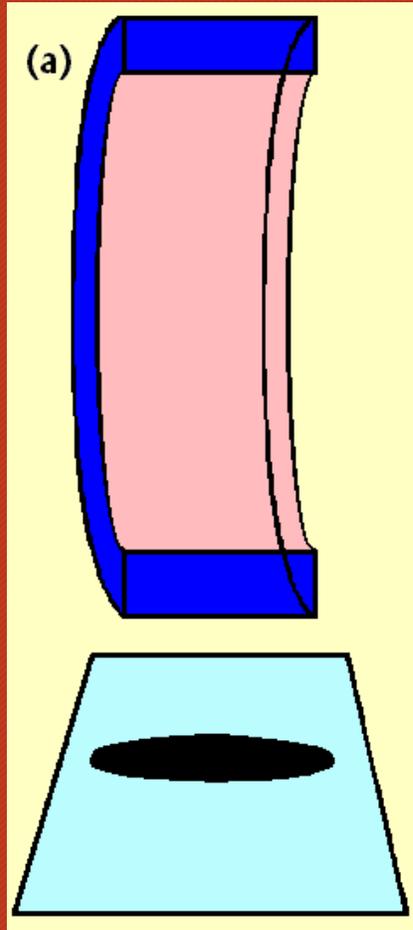


Image quality of x-ray maps

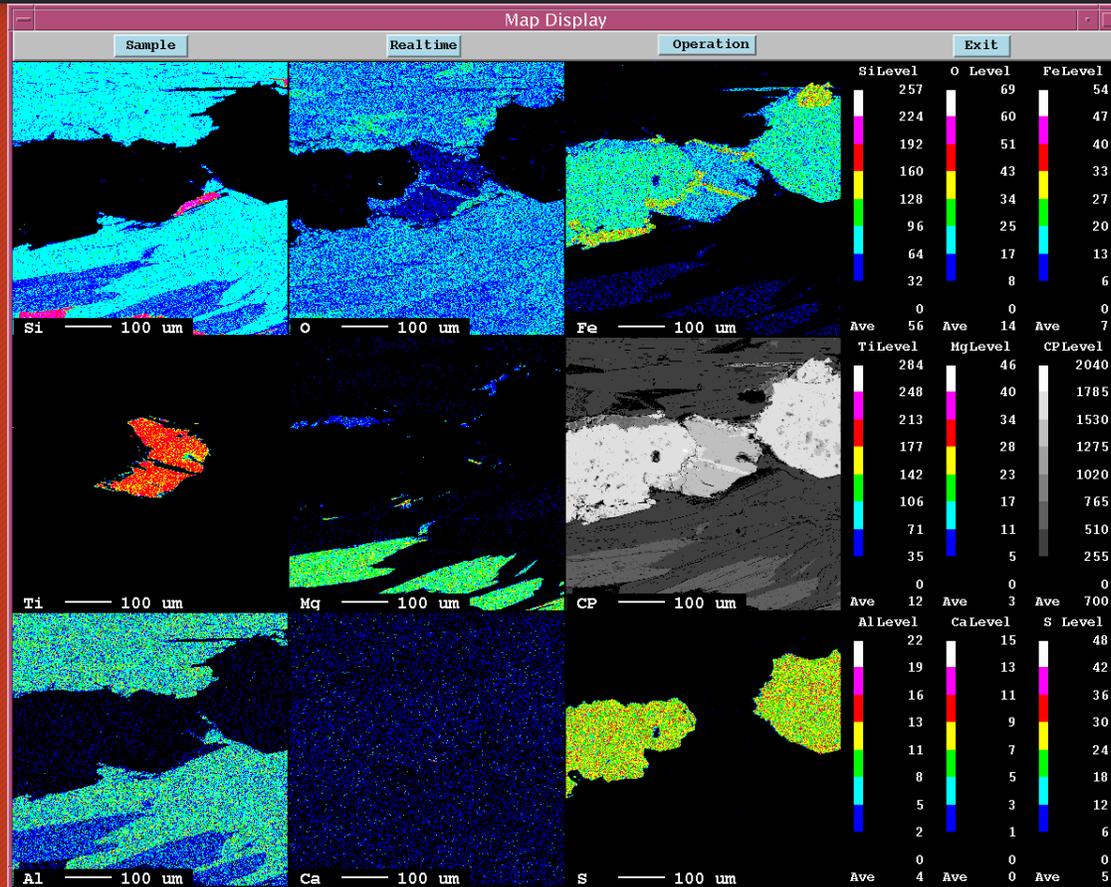
39

Two factors:

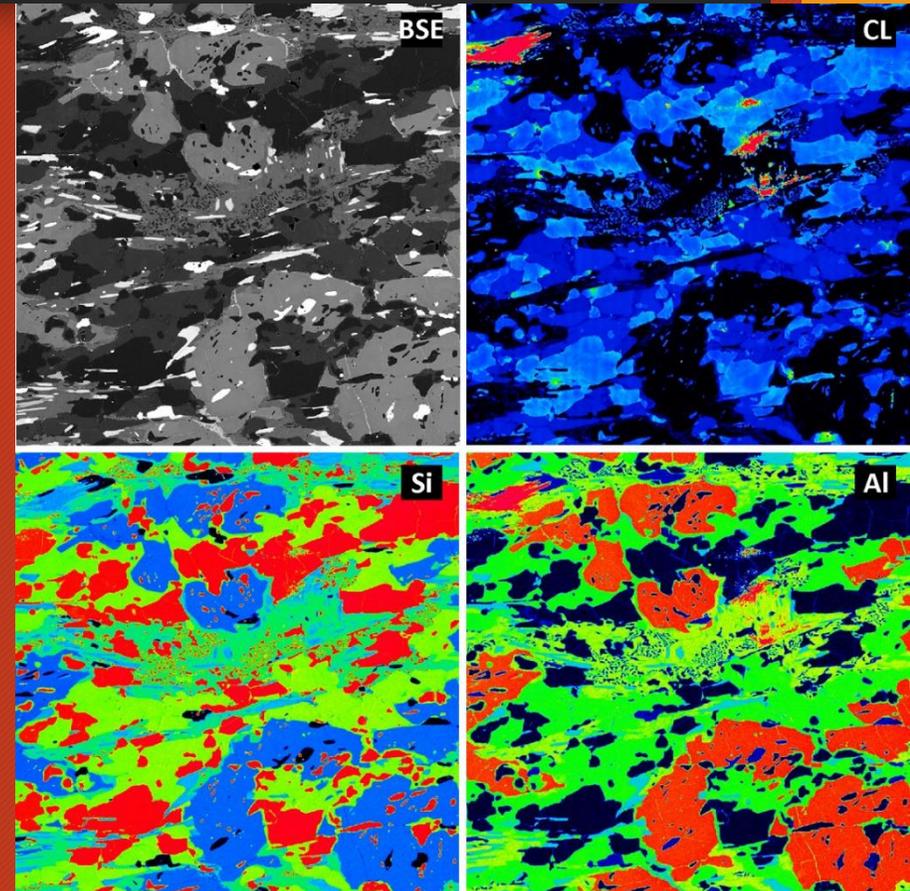
- ❑ Image resolution:
number of points measured within the imaged area
- ❑ X-ray Signal:
beam current and counting (dwell) time per point

Simultaneous mapping with different signals

40



Combined BSE, WDS and EDS X-ray mapping



Combined BSE, CL and X-ray mapping